

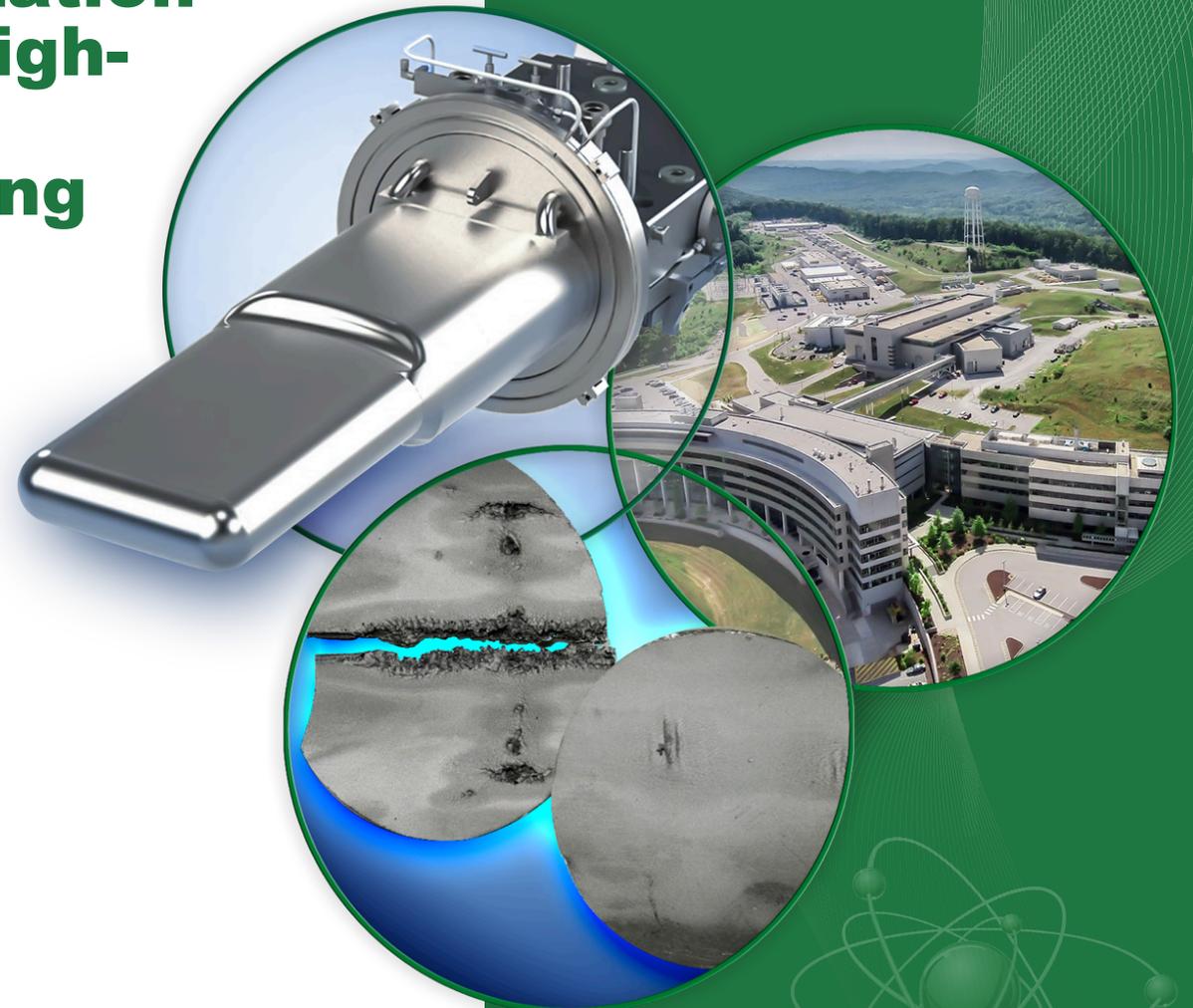
# Tensile properties characterization of irradiated AISI 316L from high-use target modules at the Spallation Neutron Source using digital image correlation

David McClintock  
Maxim N. Gussev  
Cody Campbell  
Frank Garner

November 1, 2016

13<sup>th</sup> International Workshop on  
Spallation Materials Technology

Chattanooga, Tennessee, USA



# Outline

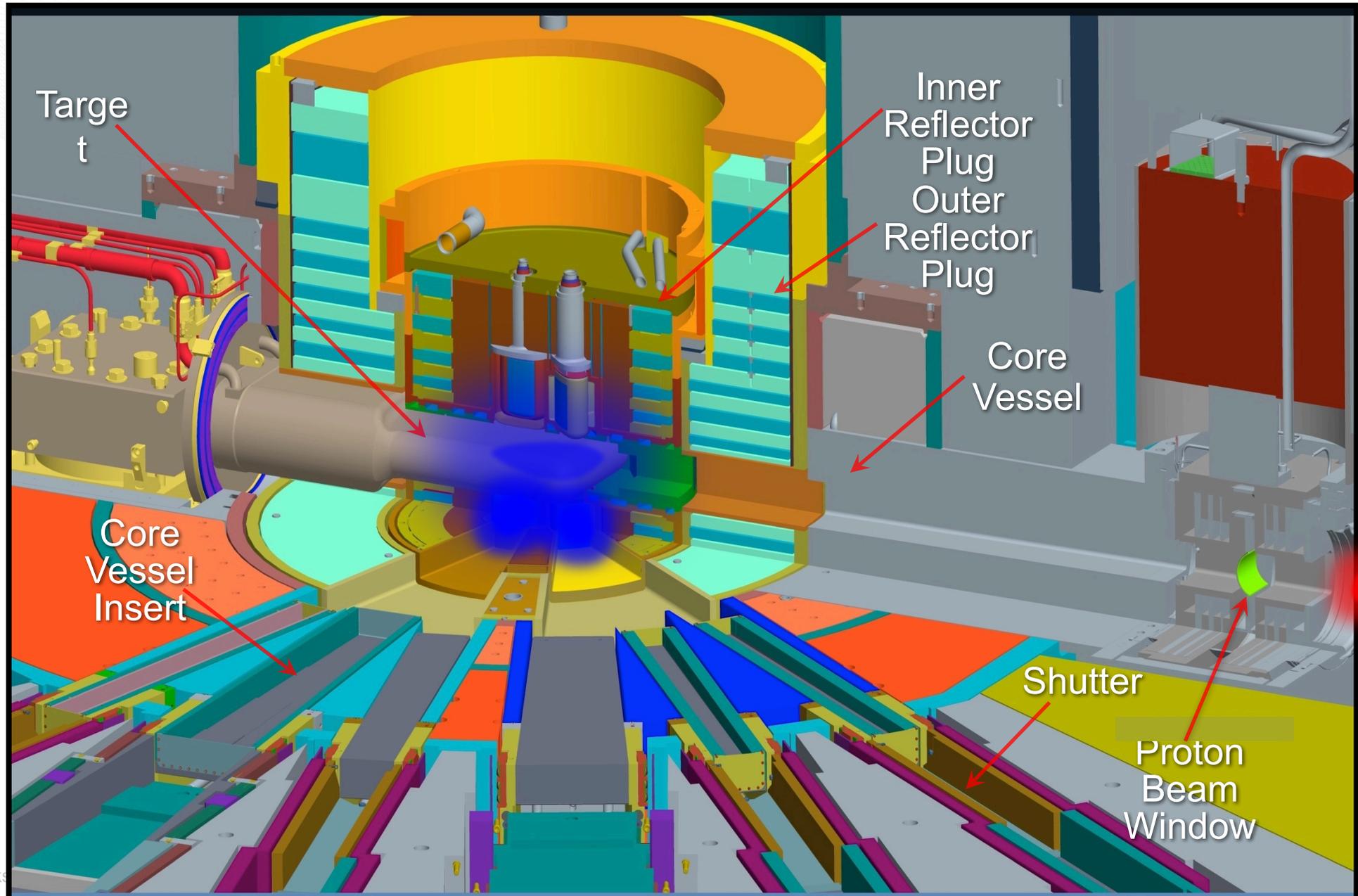
- Background on SNS Target and Previous Work
- Target Sampling and Testing Procedures
- Results
  - Tensile Testing
  - Digital Image Correlation Analysis
- Summary

# The Spallation Neutron Source is a megawatt class accelerator-based pulsed neutron source

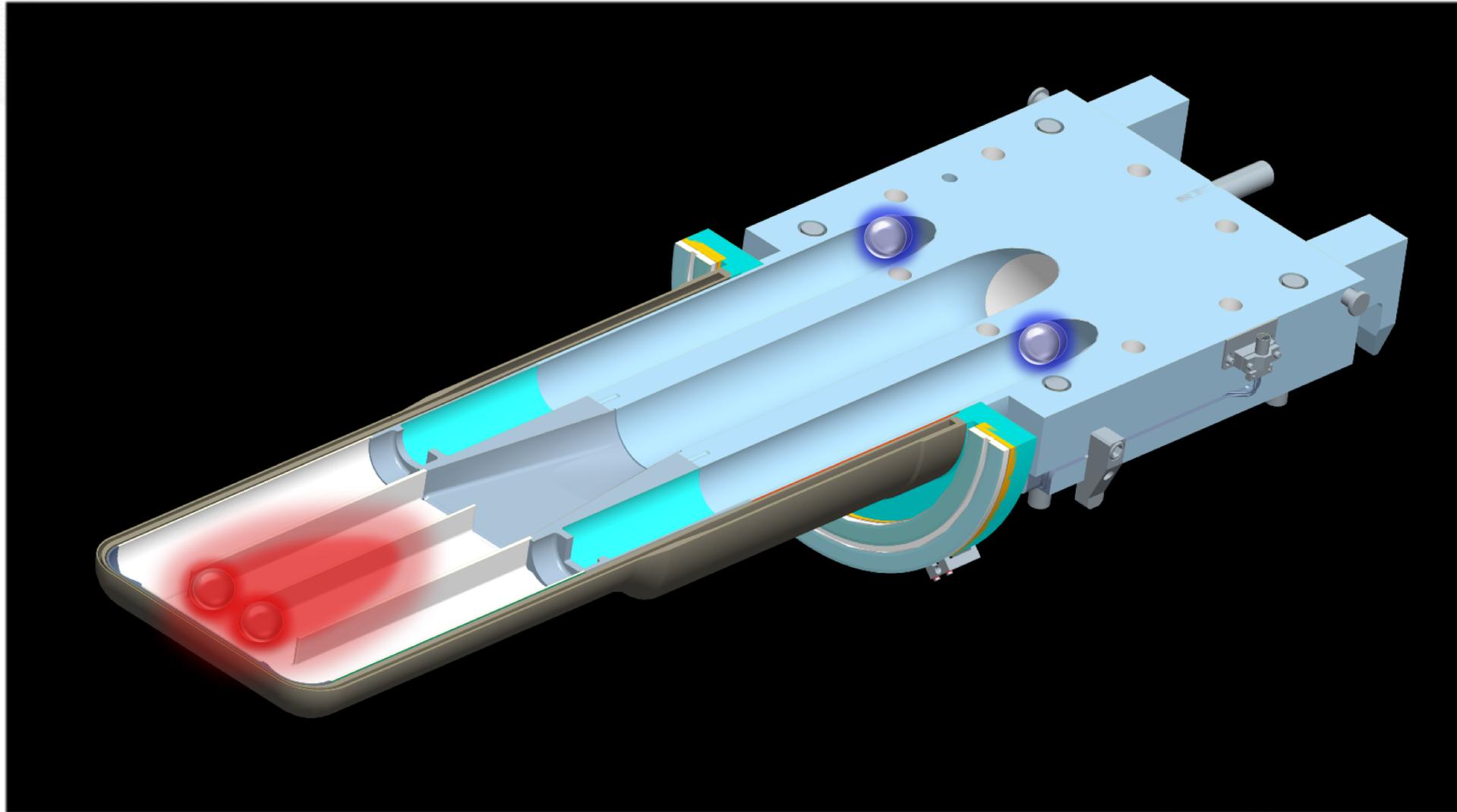
- Neutrons are produced via high-energy spallation reactions induced by injecting 1 GeV protons into liquid mercury at a frequency of 60 Hz



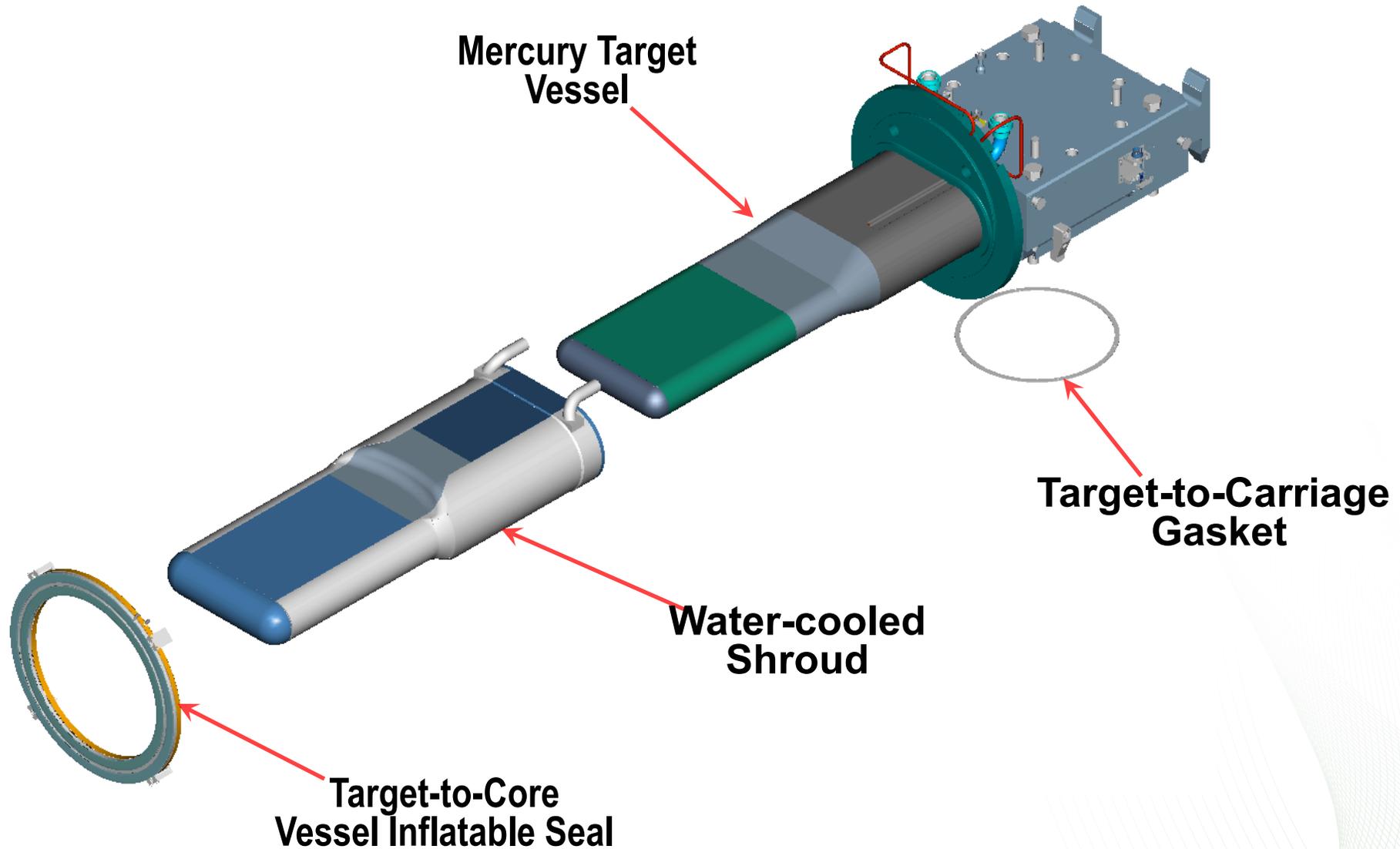
# The target provides neutrons to 24 beam lines



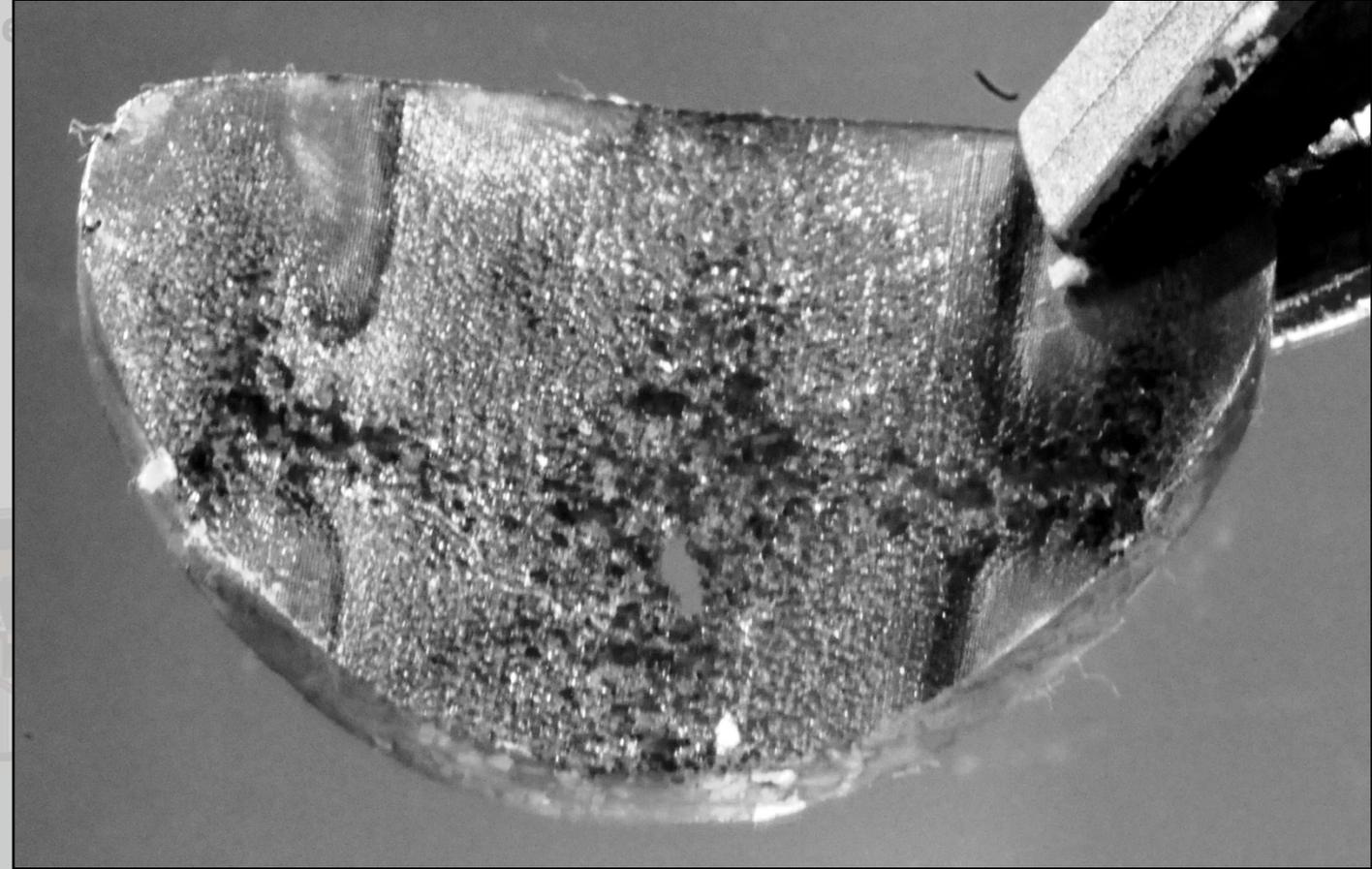
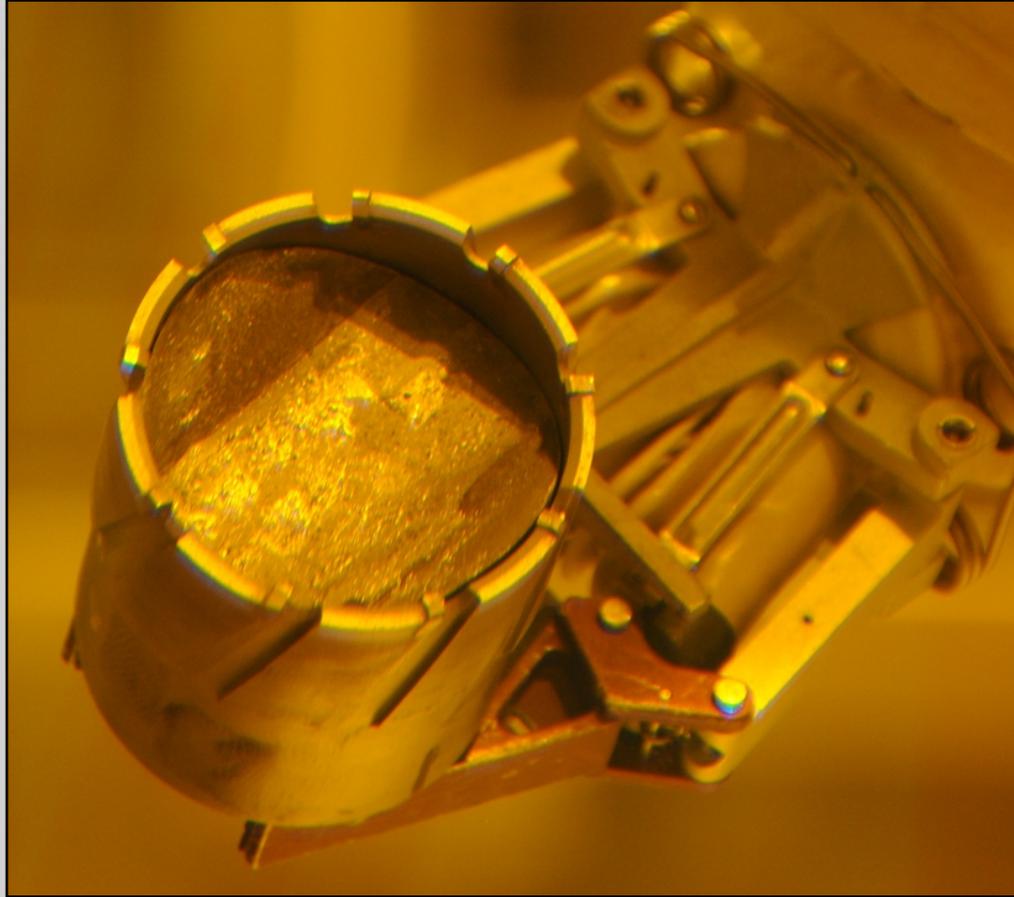
# Mercury enters through side supply passages and returns through the center return passage



# SNS target is composed of two “vessels” welded to a manifold block

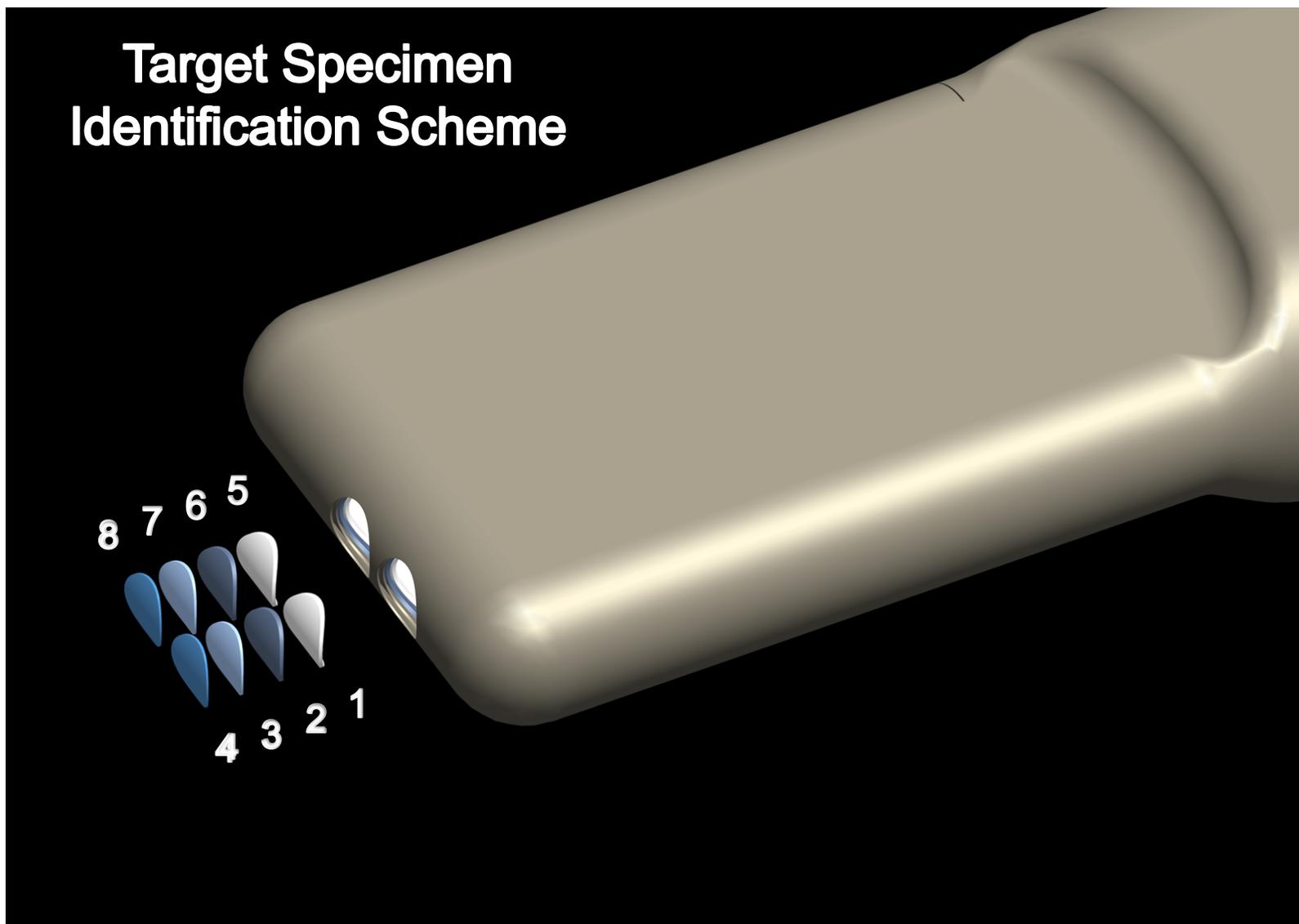


# Disk-shaped Specimens are Routinely Sampled from the Beam Entrance Region of SNS Targets After Service



Pneumatic  
Piston

# Four disk-shaped specimens were removed from the target module by each cutter



# The SNS has secured several contracts with BWXT in the past to perform detailed PIE characterizations on target specimens

- The disks were cleaned in an ultrasonic bath and photographed; the images were used to produce specimen maps for each disk
- Specimen machining maps were produced for each disk and test specimens were machined via electrical-discharge machining (EDM)
- Several characterization techniques have been employed, including: tensile testing, hardness testing, SEM examination of cavitation pitting, surface replication/SEM examination, and inclusions characterization

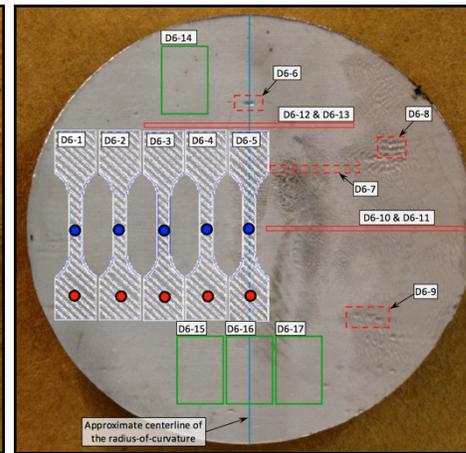
## Example: Disk 6 from Target 2



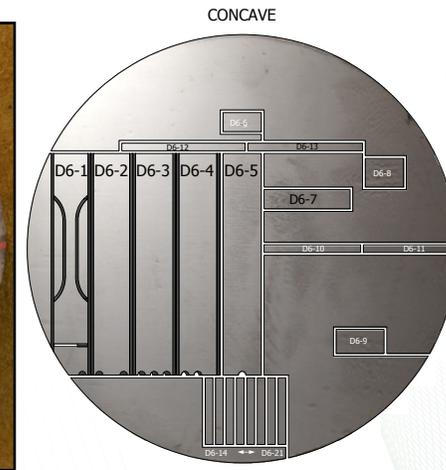
**Before Cleaning**



**After Cleaning**



**Specimen Map**

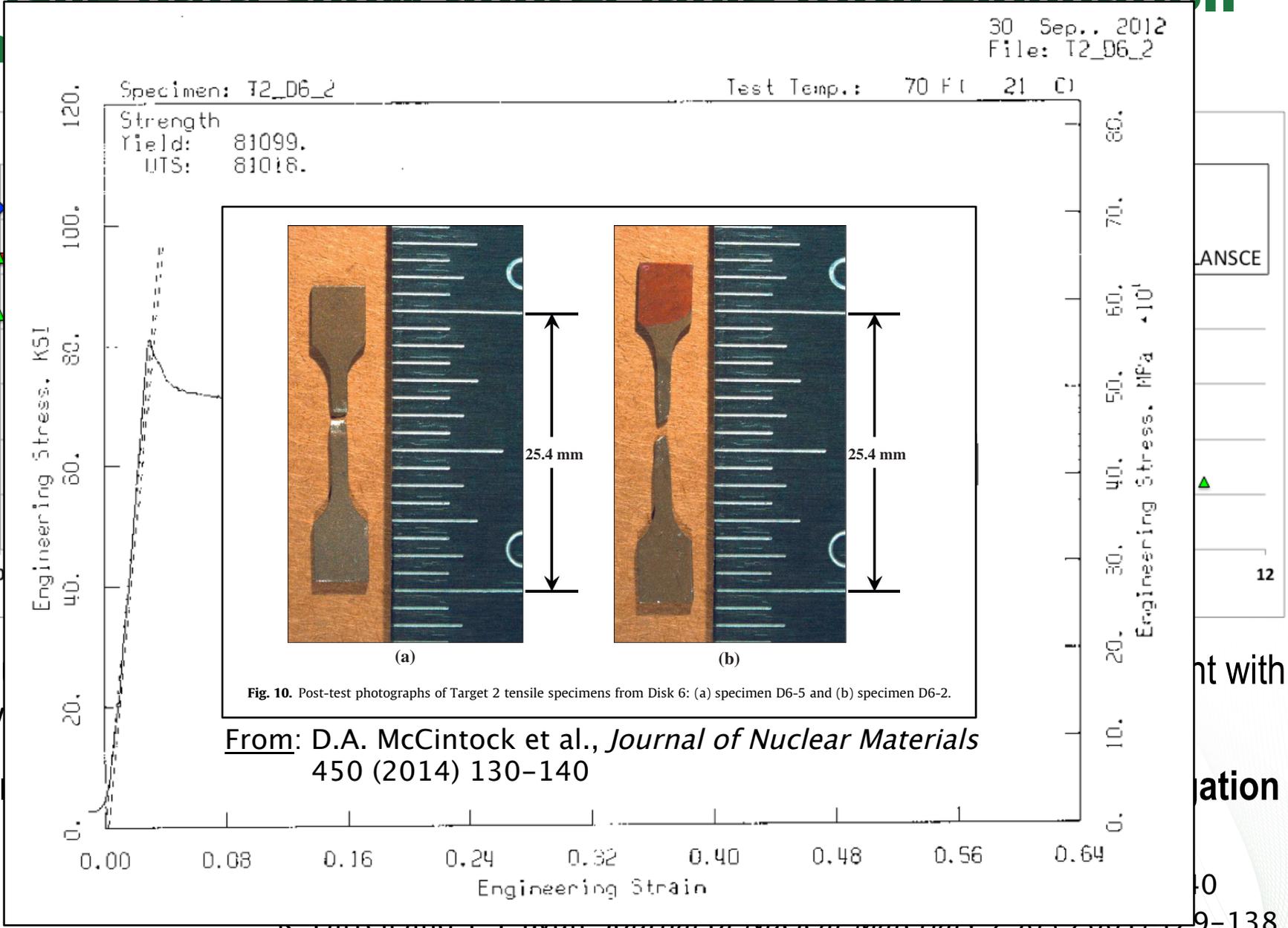


**EDM Machining Map**

# Tensile data show appreciable total elongation

- Red
- App

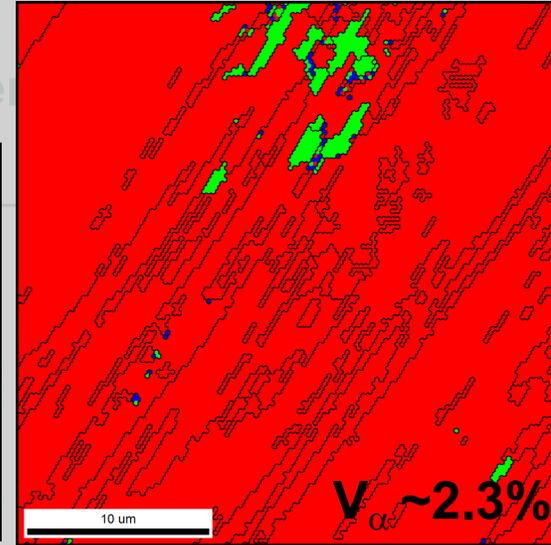
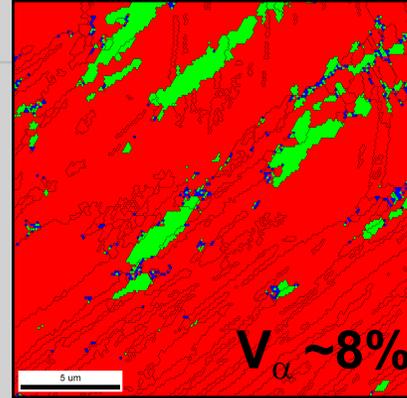
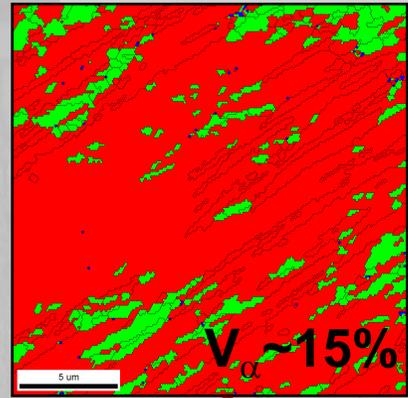
Figures



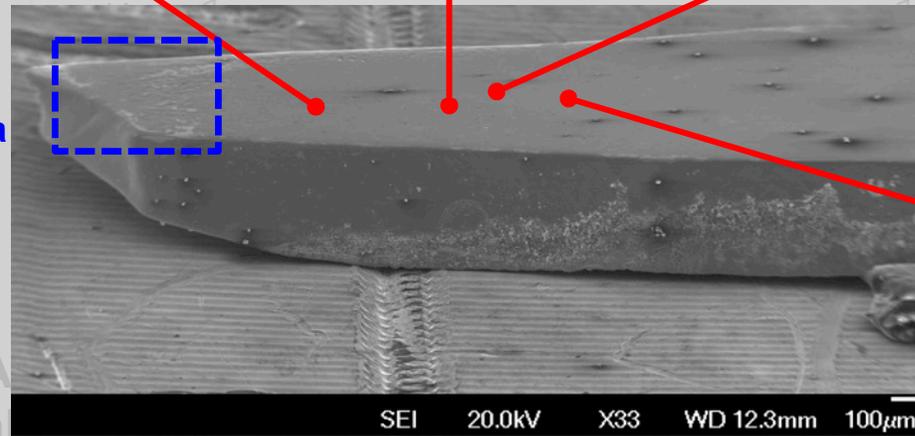
# Inability to neck produces a “deformation wave” in highly irradiated stainless steel\*

\*Slide Courtesy of: Frank Garne

For all maps:  
 $\gamma$ : Red;  
 $\alpha$ : Green;  
 $\epsilon$ : Blue.



No successful scans for this area



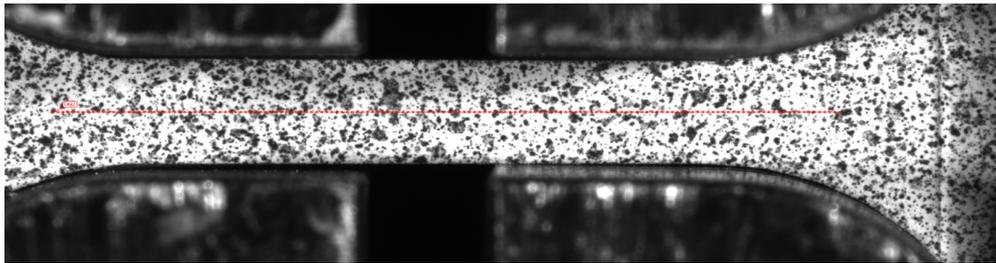
Scan #7, 2000x



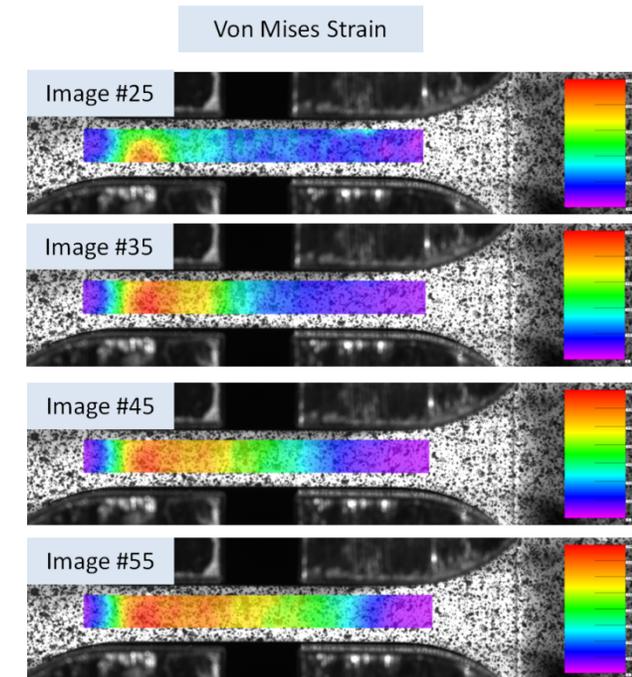
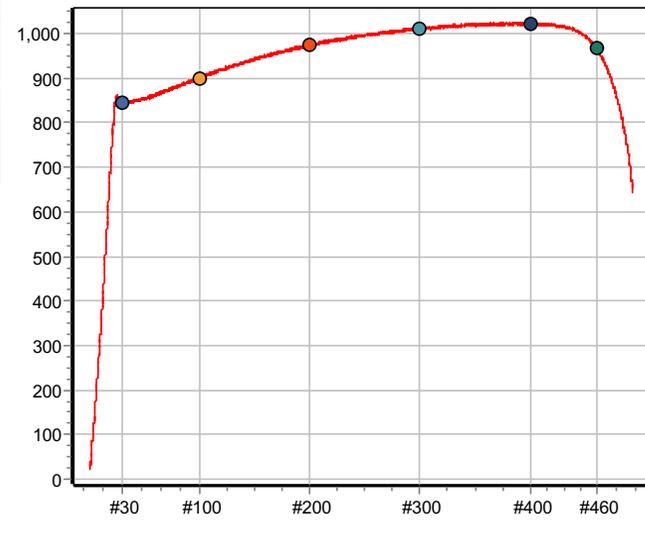
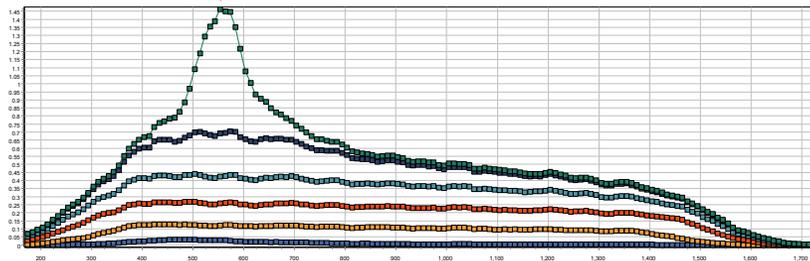
Gusev et al., J. Nucl. Mater. 386–388 (2009) 273–276  
Gusev et al., J. Nucl. Mater. 403 (2010) 121–125

# Digital image correlation (DIC) is an extremely powerful tool for analyzing deformation behavior

- DIC allows for analyzing deformation localization in detail, calculating strain distribution along the specimen gauge, strain rate maps, etc.



Neck Formation



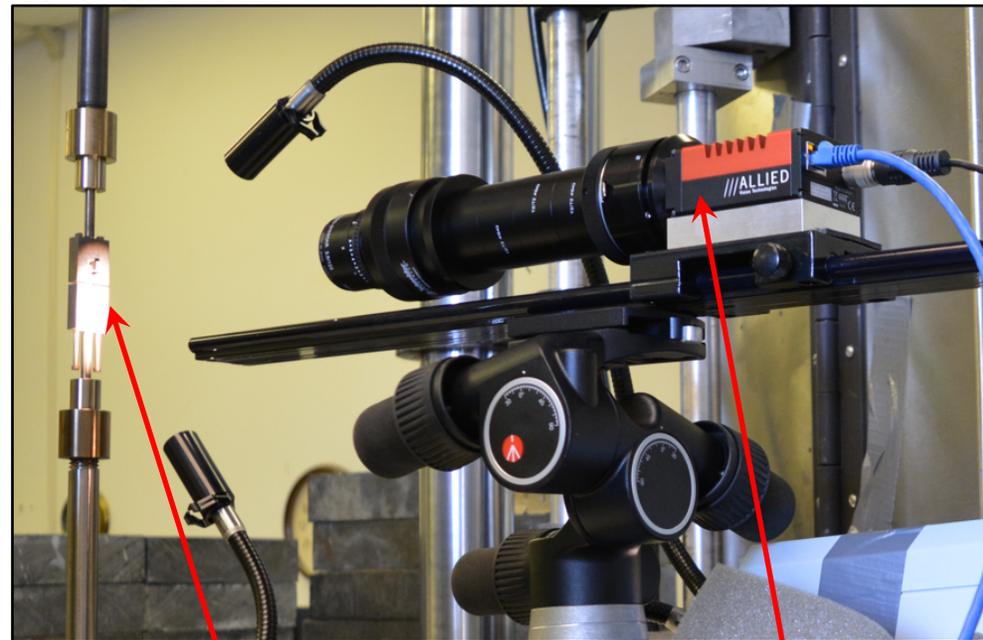
- Strain distribution (von Mises strain, Green-Lagrange strain tensor)
- Subset 63 pix. Raw data! No smoothing/manipulations.

- **32 irradiated specimens were tested using DIC – at the moment, the largest dataset for the irradiated austenitic steels**

# Tensile specimens were prepared for DIC characterization

- Digital Image Correlation (DIC) was used to characterize specimen deformation behavior during testing
  - High-resolution camera captured one image per second during test
  - Image data reveals strain behavior of the entire specimen during test
- Tensile specimens were painted to provide contrasting features for analysis software to track and calculate displacements

DIC Testing Setup



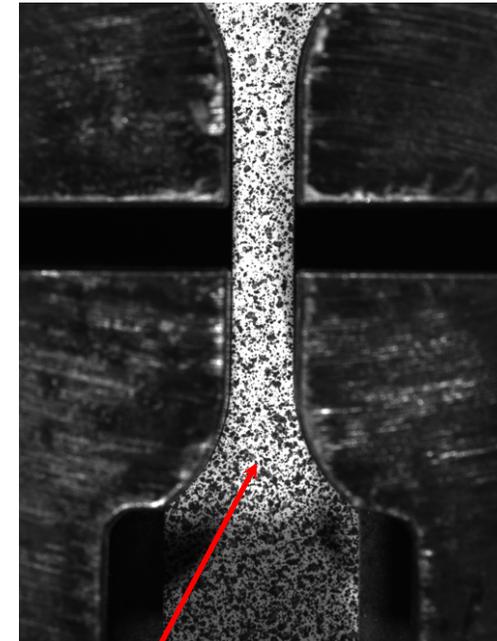
Tensile Specimen

High-resolution Camera

Target 8 Specimen

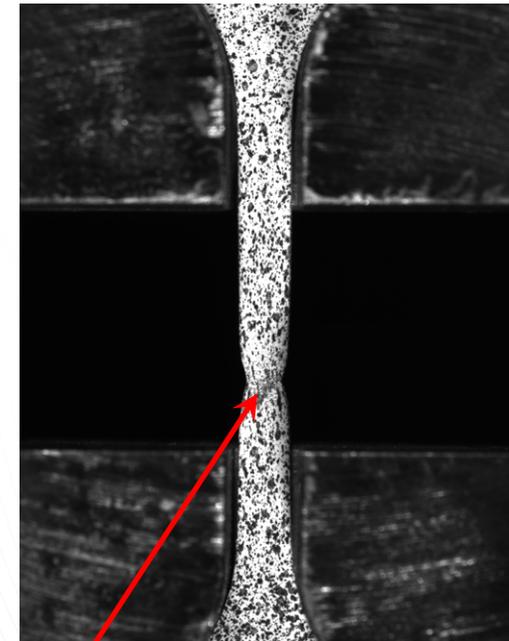


Beginning of Test



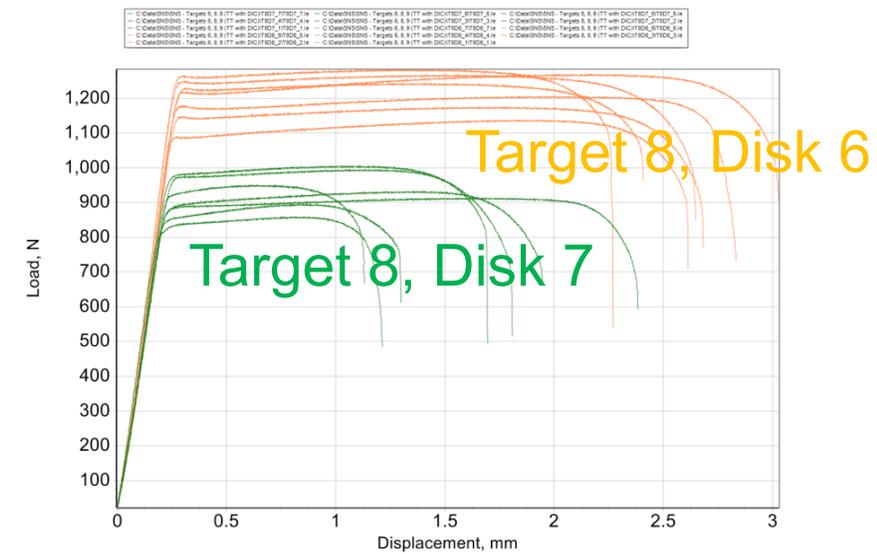
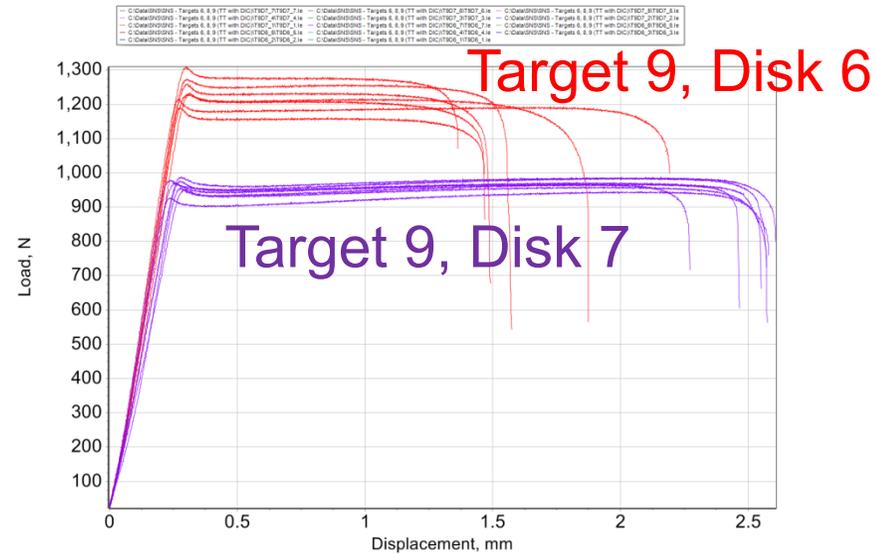
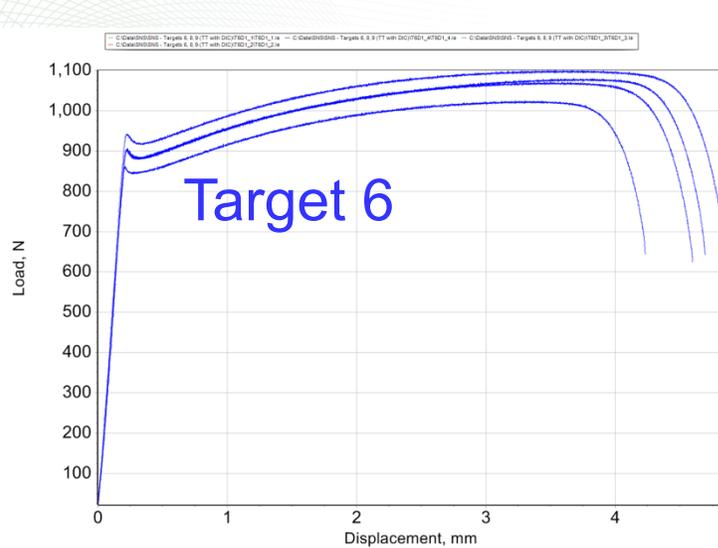
Tensile Specimen in Testing Fixture

End of Test



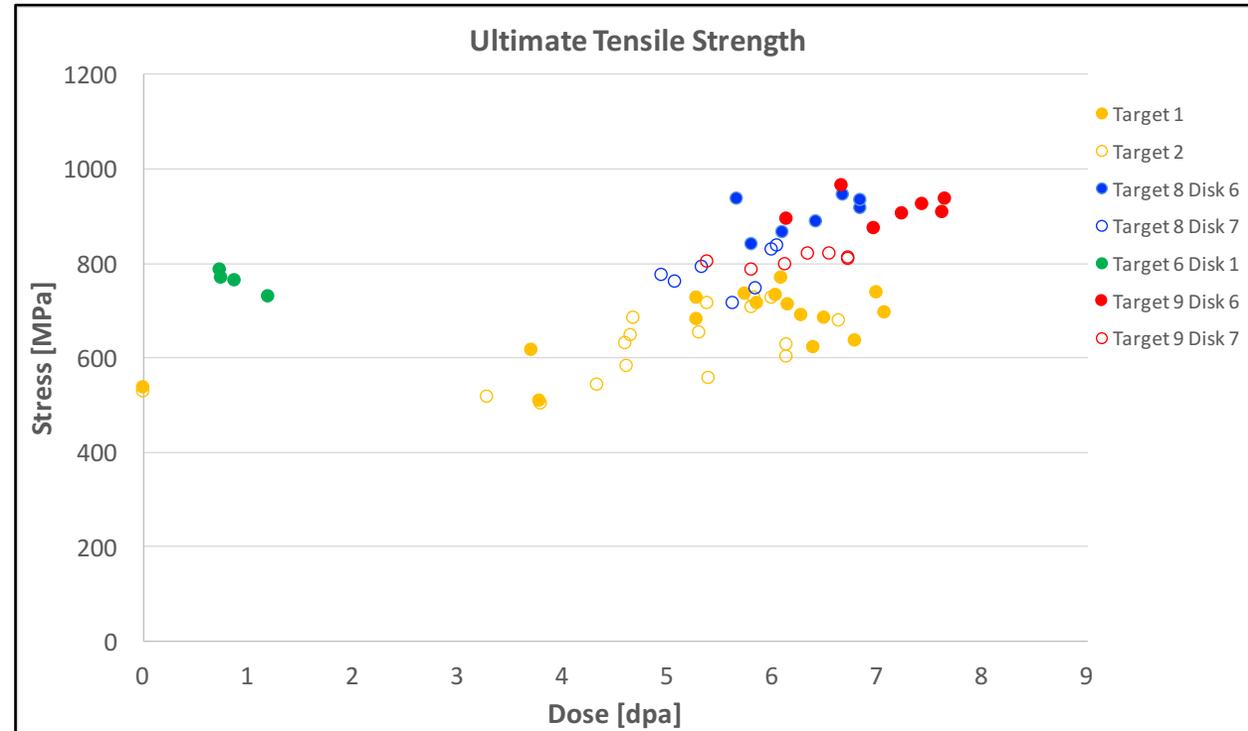
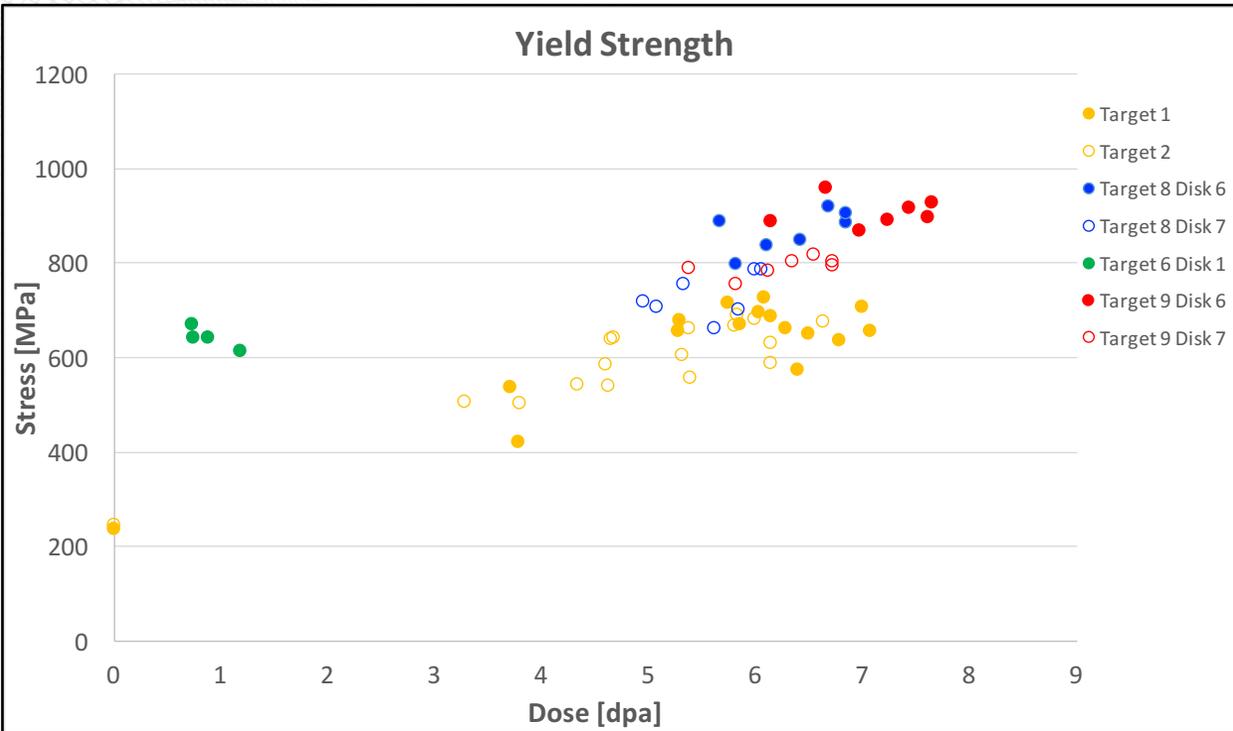
Neck at Specimen Failure Location

# Engineering stress strain curves show similar general deformation behavior



- Shapes of curves were quite similar, with a load drop shortly after yield
- Only Target 6 specimens (0.75 – 1.2 dpa) had appreciable strain-hardening after yielding

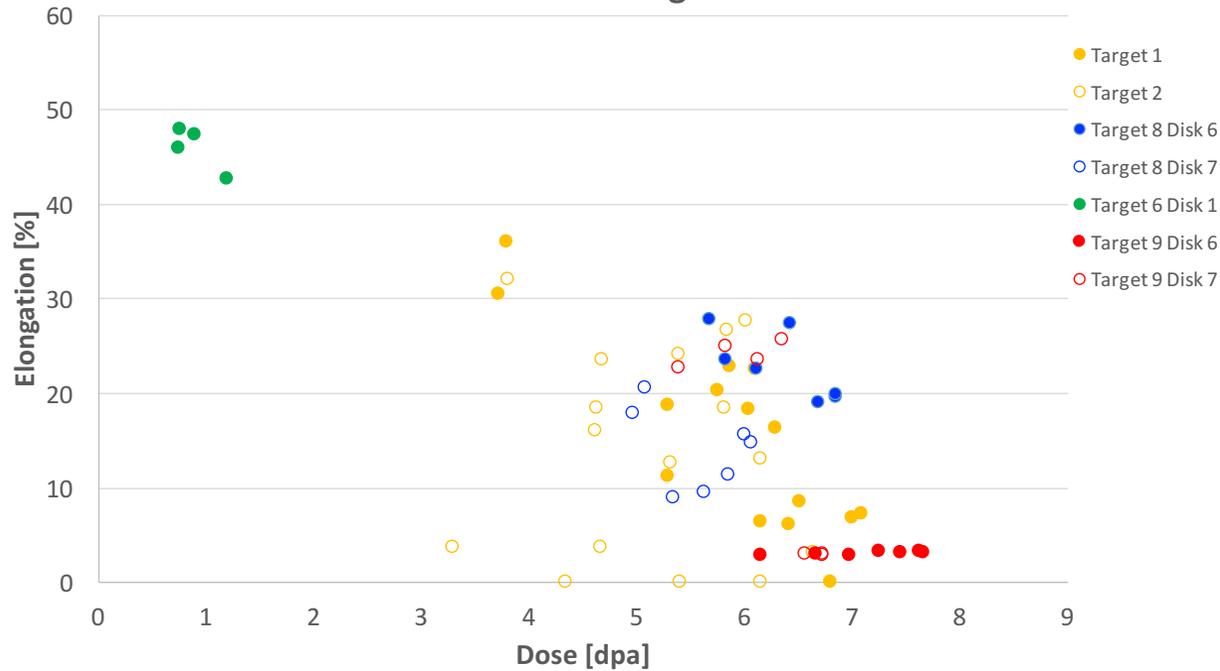
# Increases in yield and ultimate strengths were observed



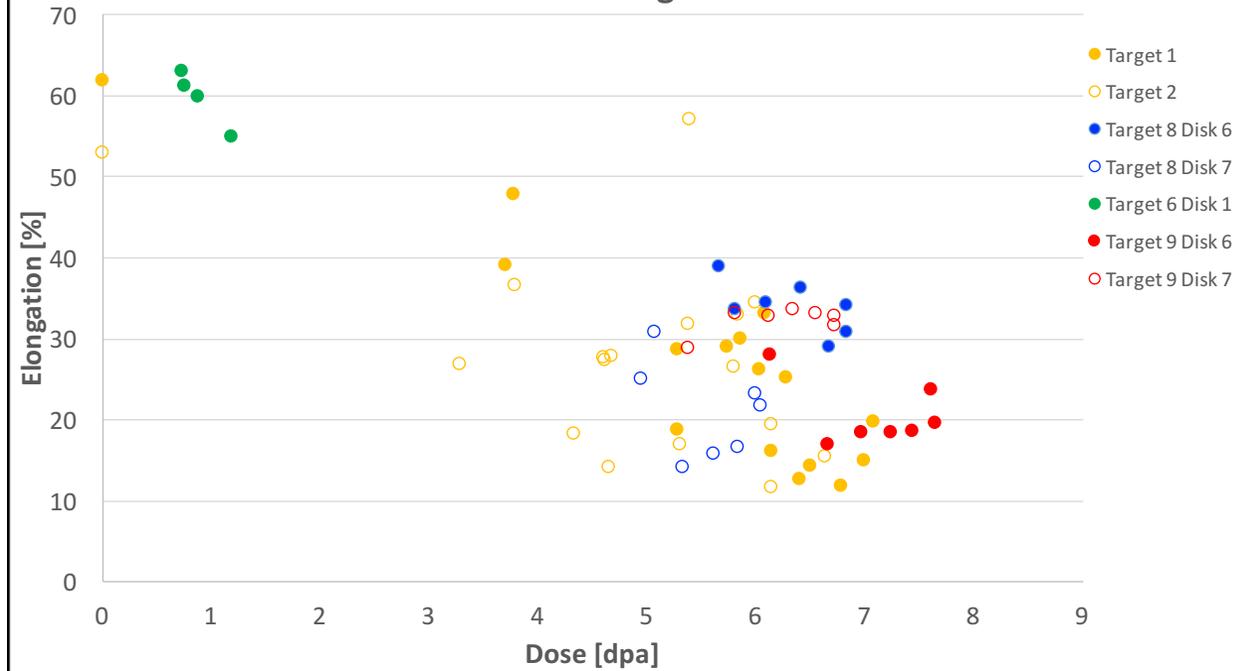
- Typical “low temperature” hardening was observed

# Elongation results show appreciable ductility remained in 316L target material up to ~7.6

### Uniform Elongation

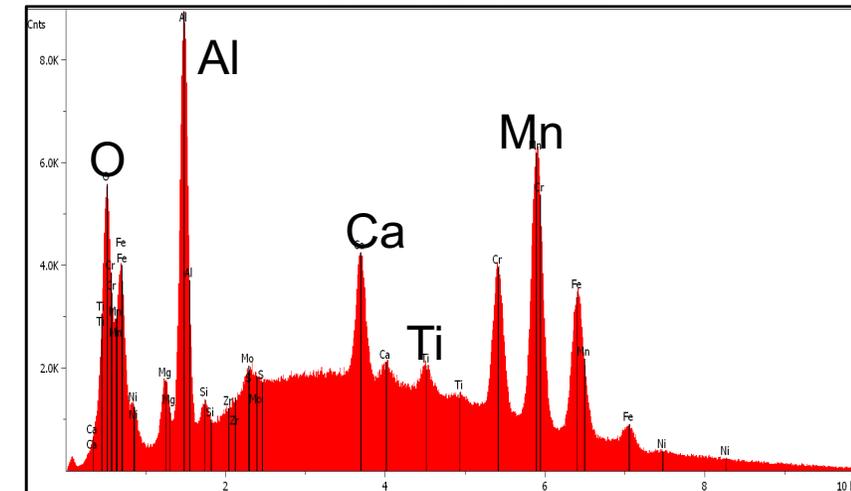
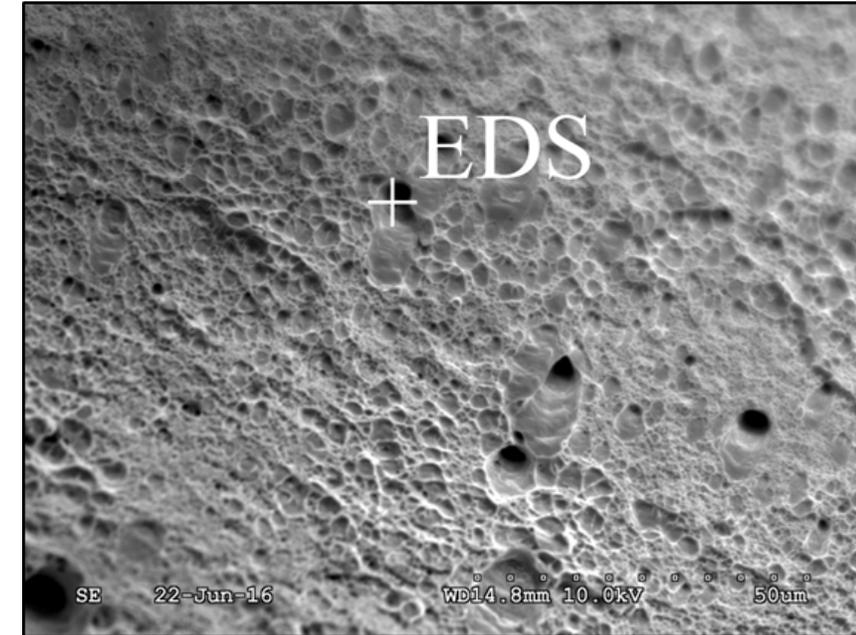
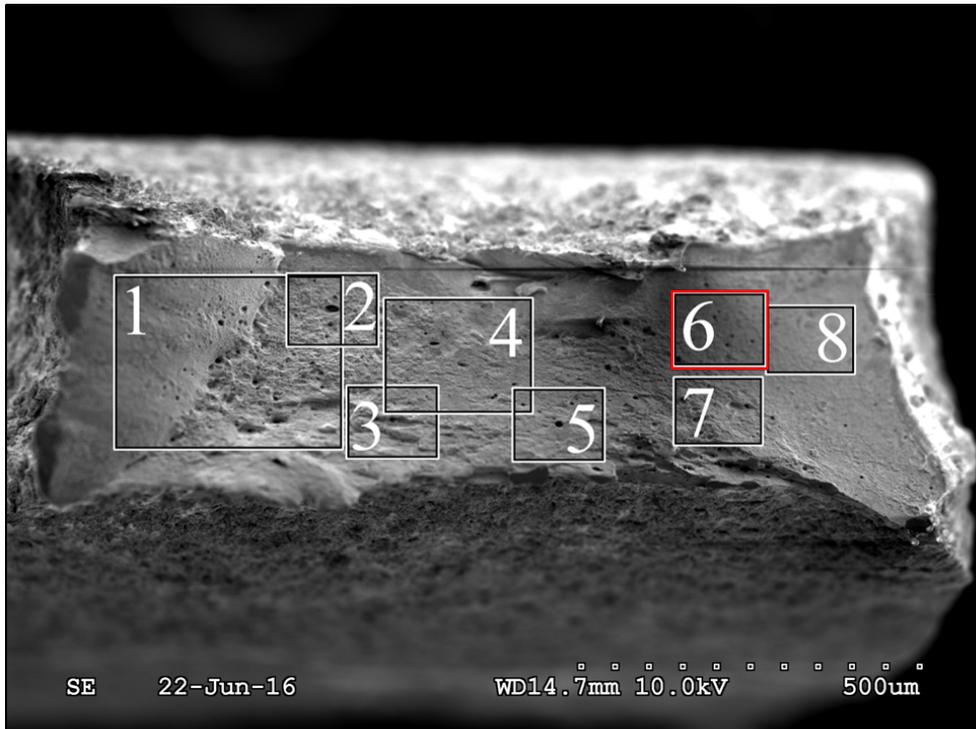


### Total Elongation



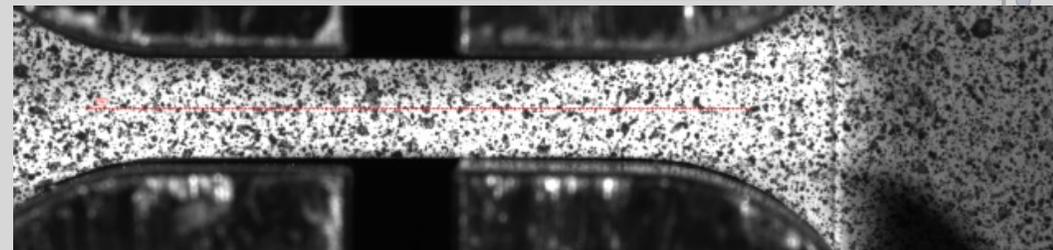
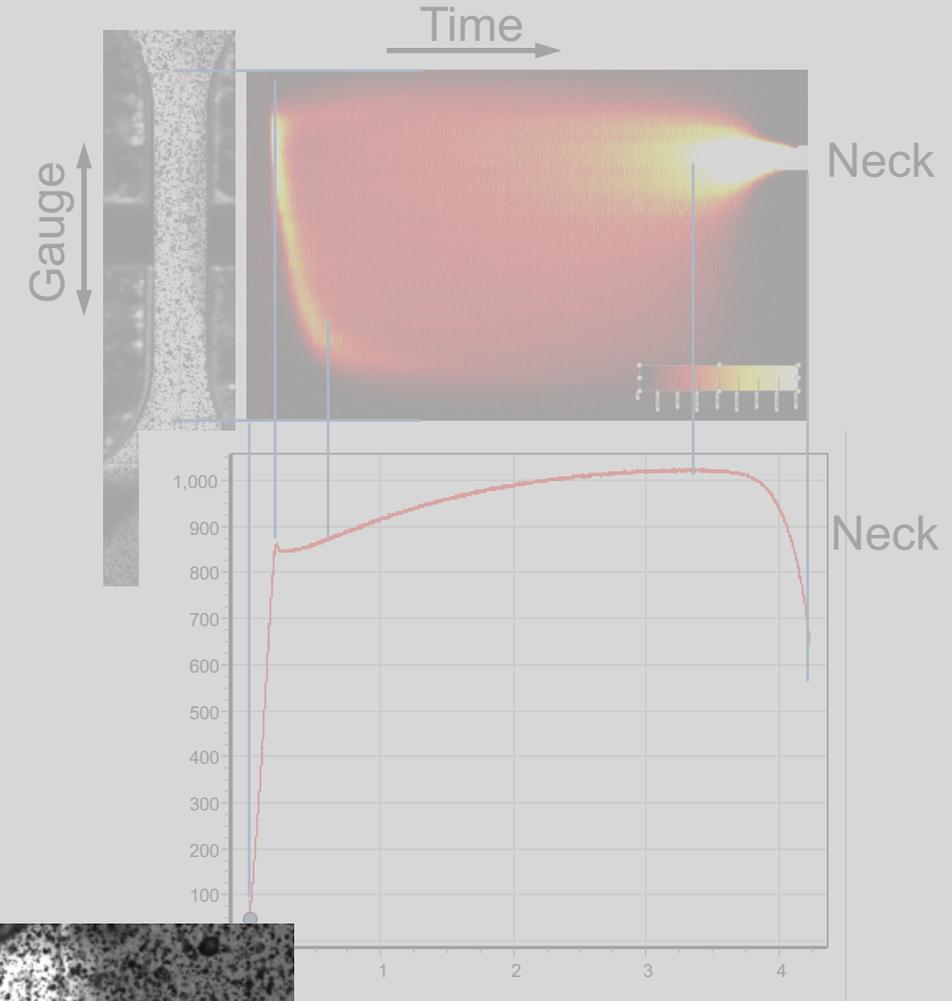
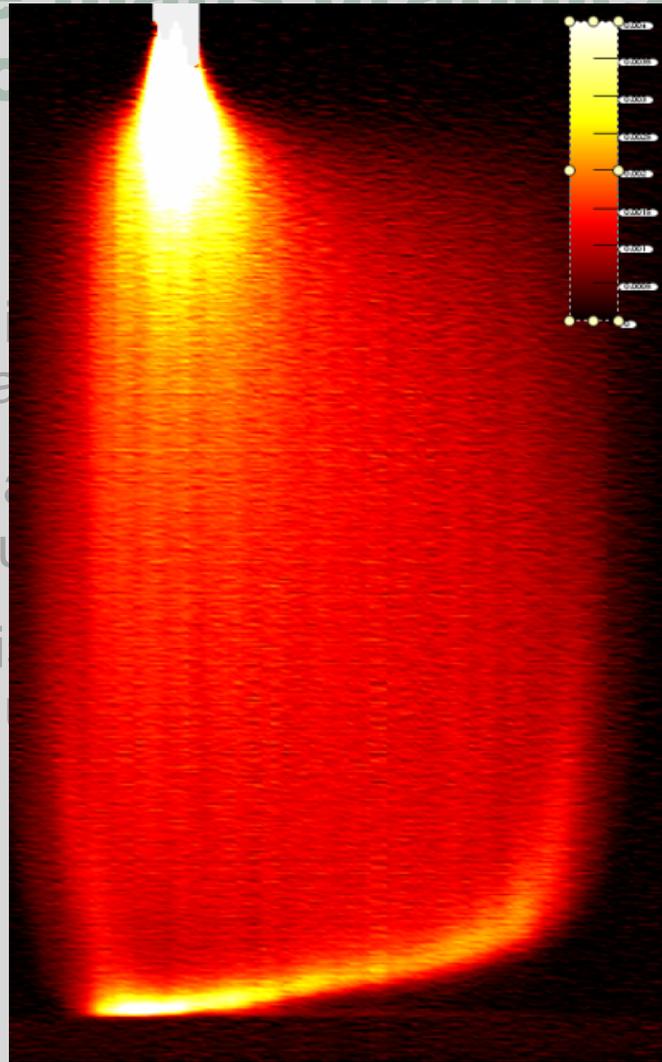
# Fracture surfaces were examined via scanning electron microscopy

- All specimens fracture surfaces were primarily ductile microvoid coalescence morphology
- Inclusions rich in manganese, calcium, aluminum, silicon and titanium were identified via EDS



# Strain rate maps graphically illustrate the deformation of tensile specimens

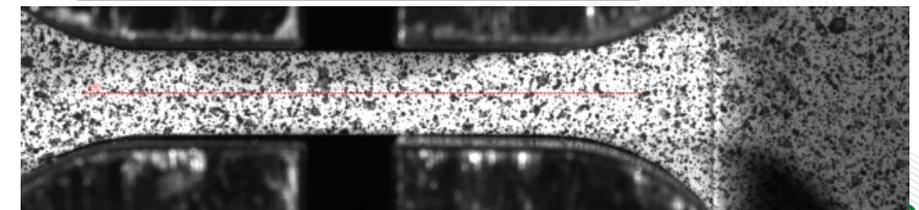
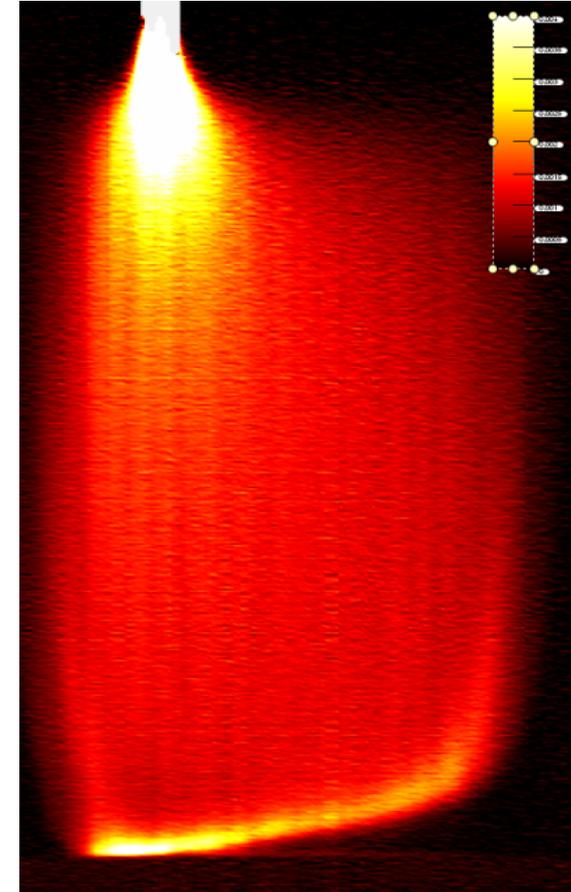
- Strain rate maps illustrate the deformation of the gauge section during testing, and the propagation of deformation bands.
- Elastic strain region is visible as a region of practically zero strain rate value.
- Neck evolution is also clear via the localized deformation in some area red.



# Specimens from Target 6 demonstrated moderate deformation band behavior

- At the beginning of test some Target 6 specimens displayed moderate deformation band behavior

Target 6, D9-1, TE = ~13%



# Target 8 specimens displayed localized and banded deformation behaviors

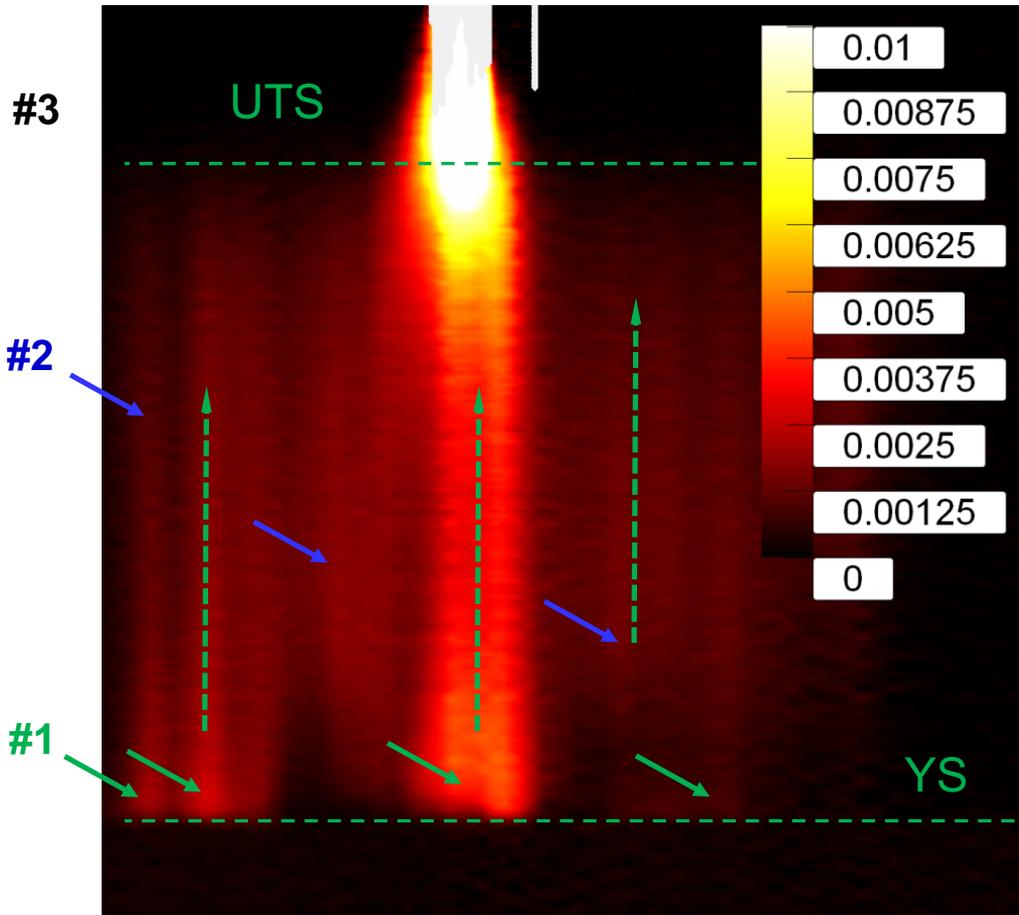
## Events:

#1. Several deformation “spots” appeared immediately after YS.

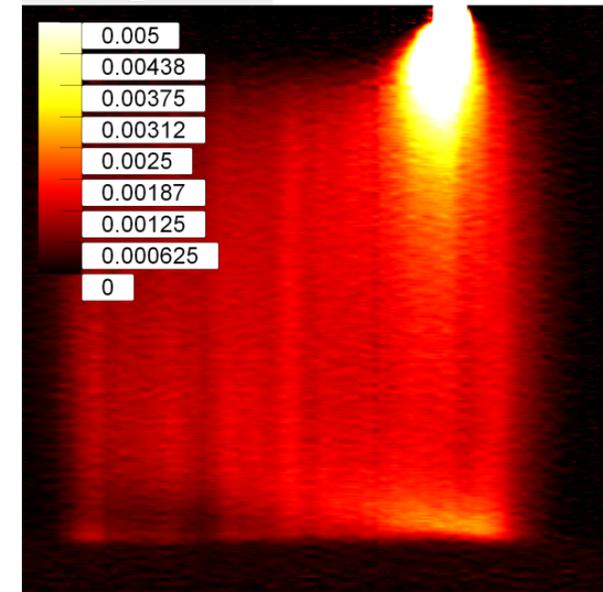
#2. “Spot” activity did not remain the same during the test. Some “spots” decayed, and new became active.

#3. At the UTS point, all active strain areas (except one) disappeared. The survived area becomes a neck.

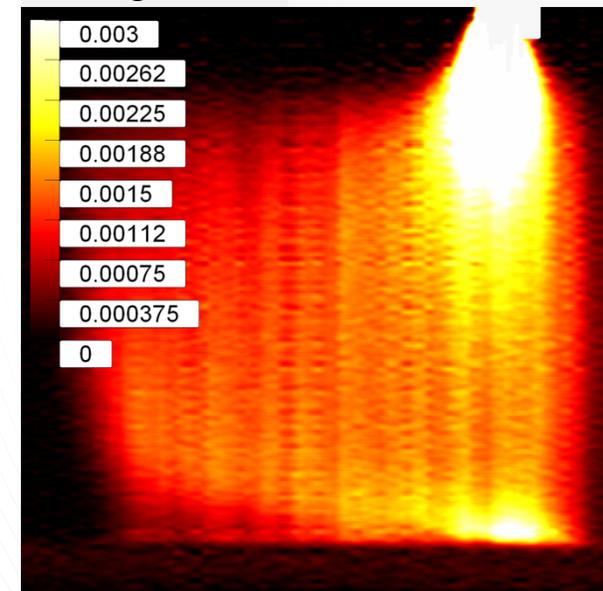
- No Luders-like band movement was observed.
- Plastic strain tended to localize in few limited areas.
- Only part of the gauge was involved in deformation resulting in TE~20-25%.



Target 8, D6-6, TE = ~36%

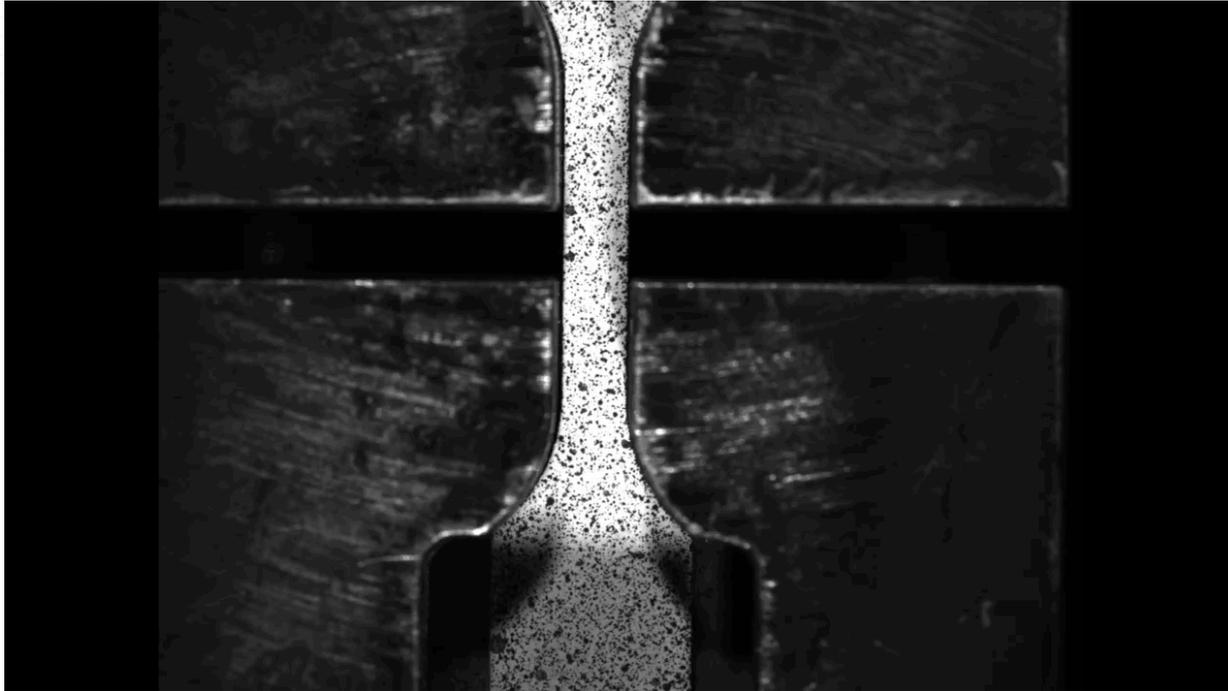


Target 8, D6-1, TE = ~39%

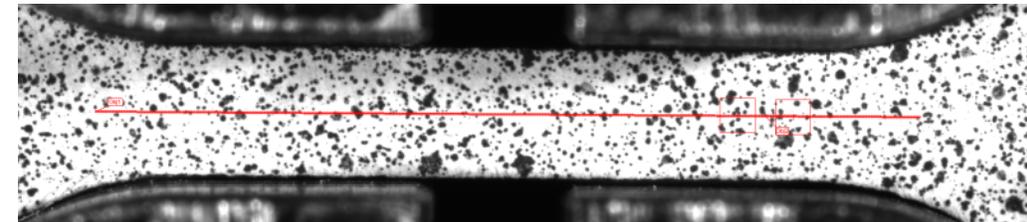
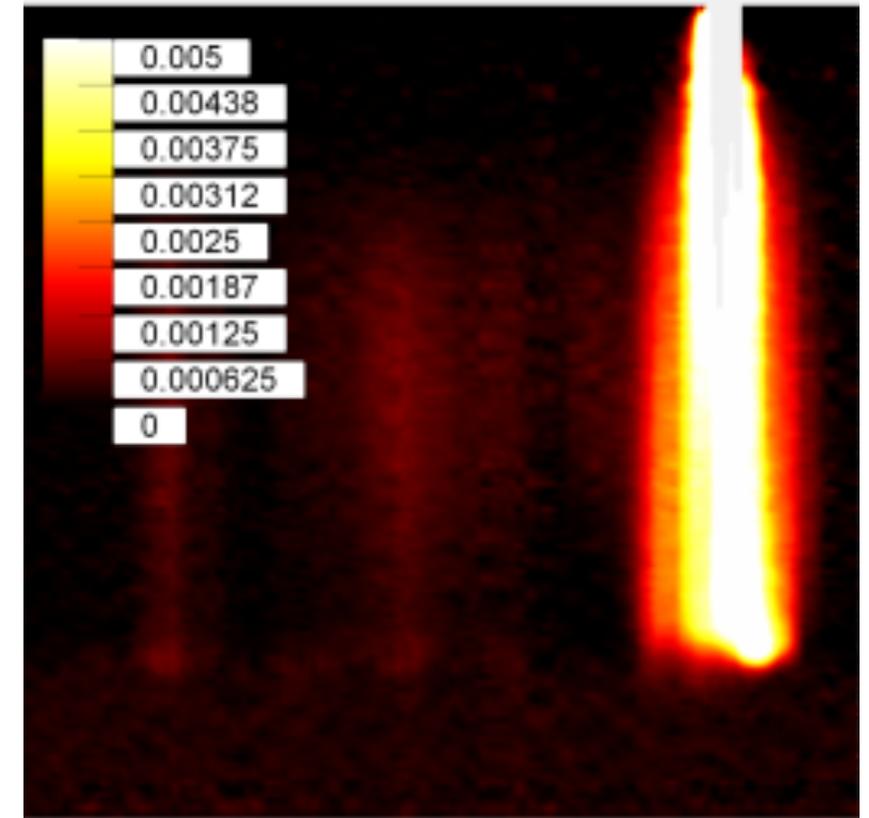


# Target 8 specimens displayed highly localized deformation

- Some Target 8 specimens displayed highly localized deformation
- Significantly reduced total elongation value

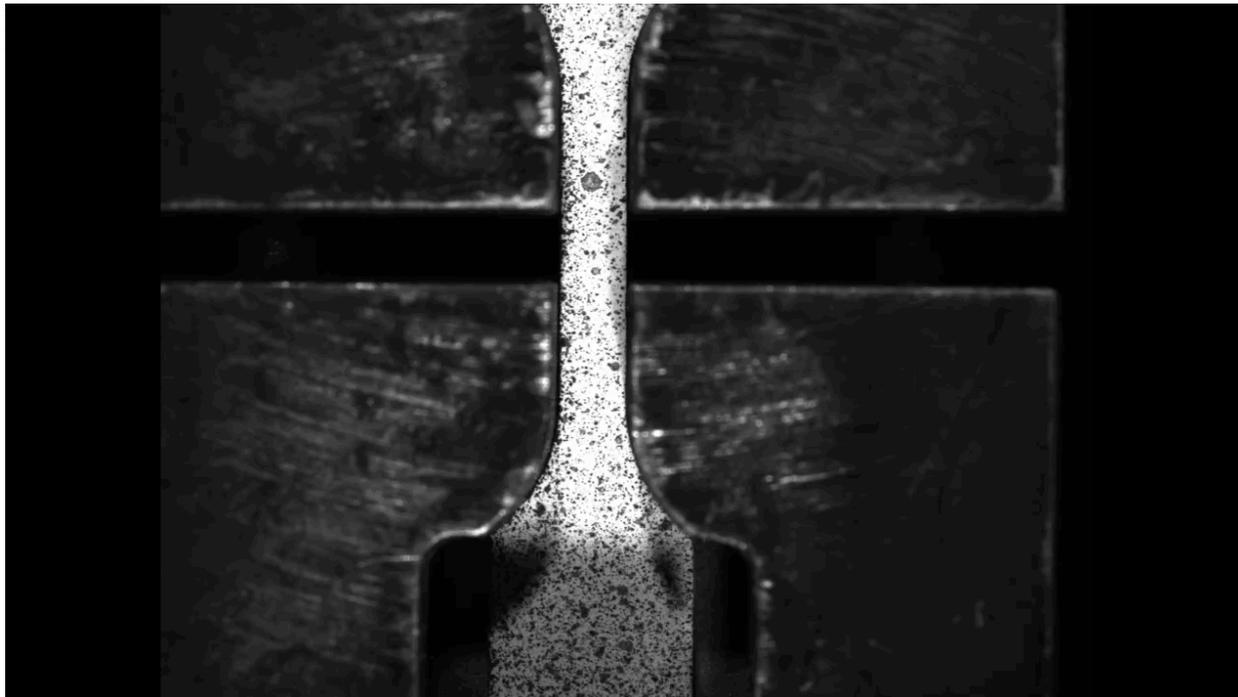


Target 8, D7-2 (5.3 dpa), TE = ~13%

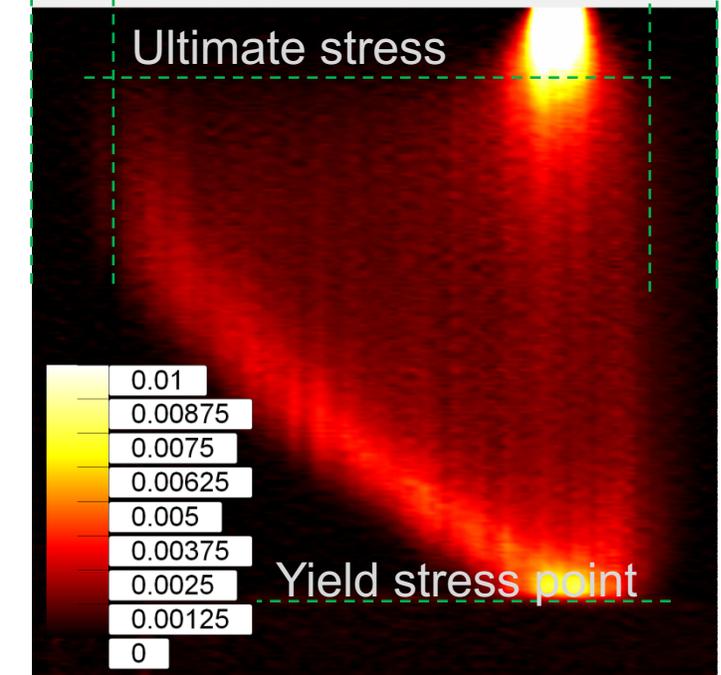
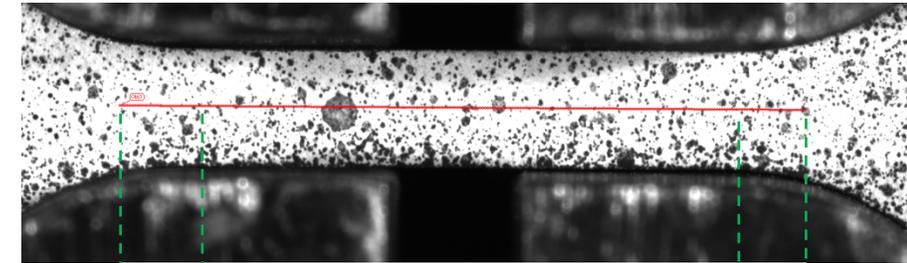


# Behavior of Target 9 specimens was varied and peculiar

- Local plastic deformation was initiated at the right side of the specimens
- Strain hardening rate was high enough to stimulate Lüders band propagation along the specimen gauge section length

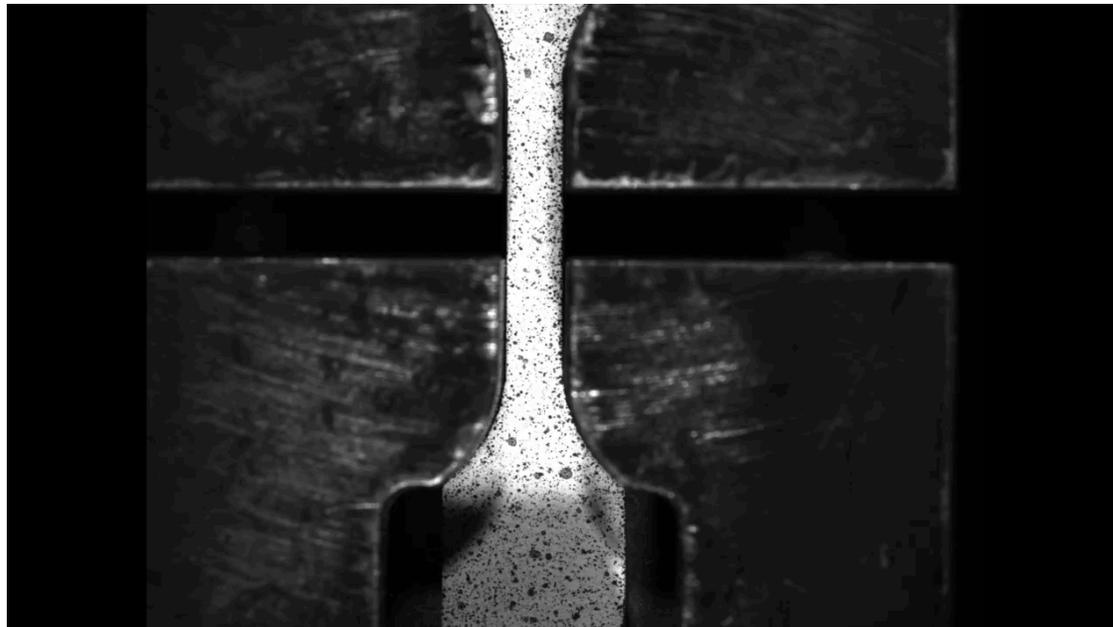


Target 9, D6-1 (6.1 dpa), TE = ~28%

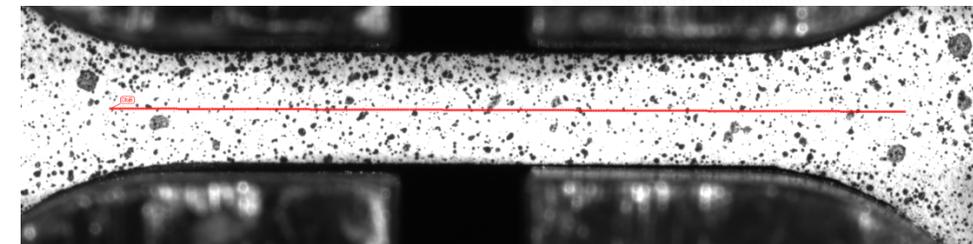
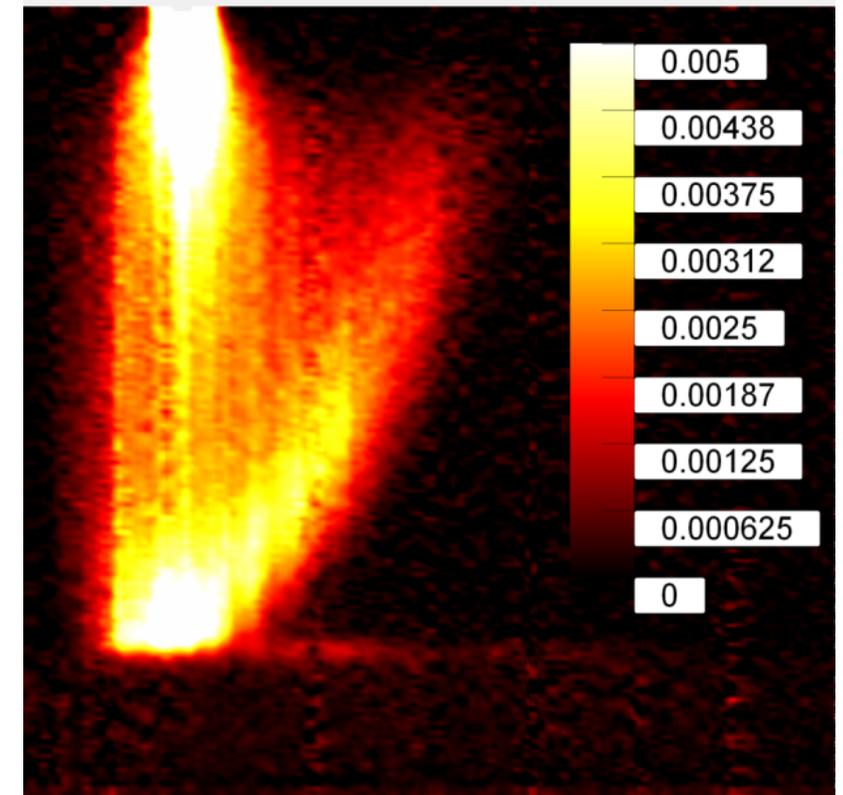


# Deformation bands on some specimens only propagated across half of gauge section

- Local plastic deformation was initiated at the left side of the specimens
- However, strain hardening rate was not high enough to support Lüders band movement
- The band propagated only part of the gauge and decreased in amplitude

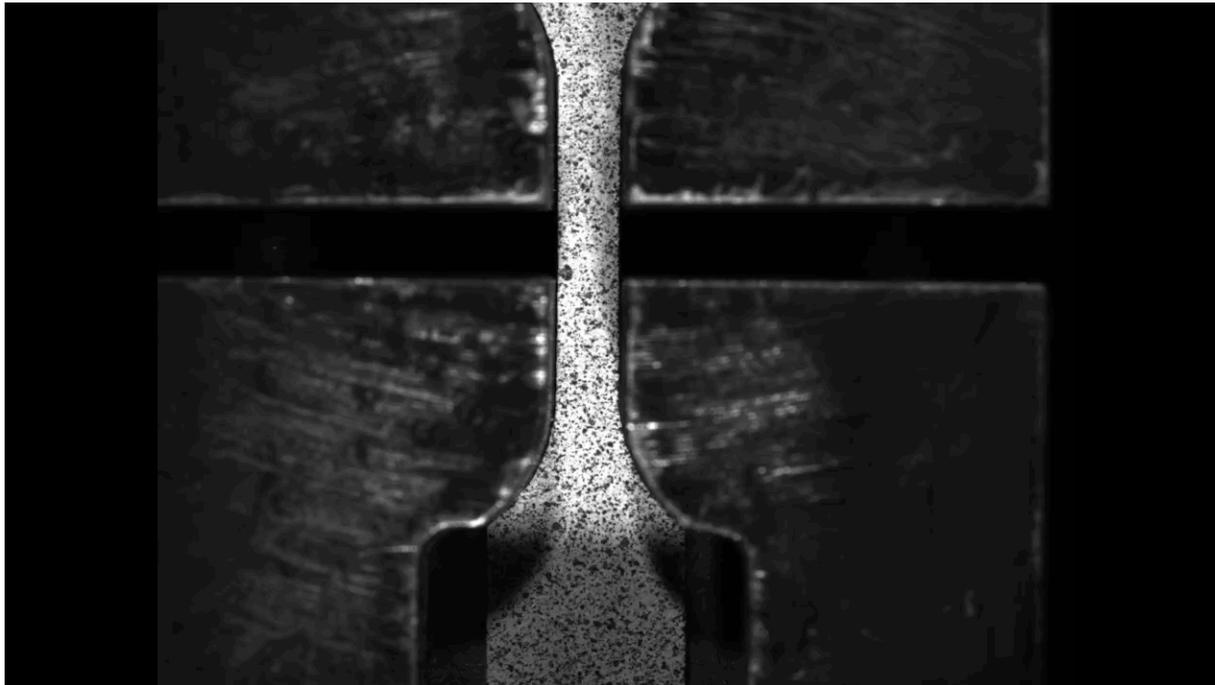


Target 9, D6-2 (7.0 dpa), TE = ~19%

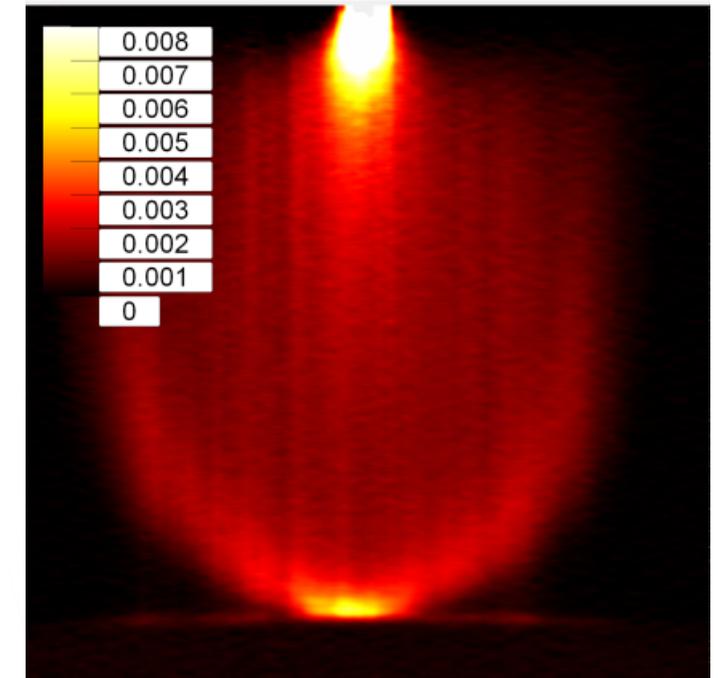
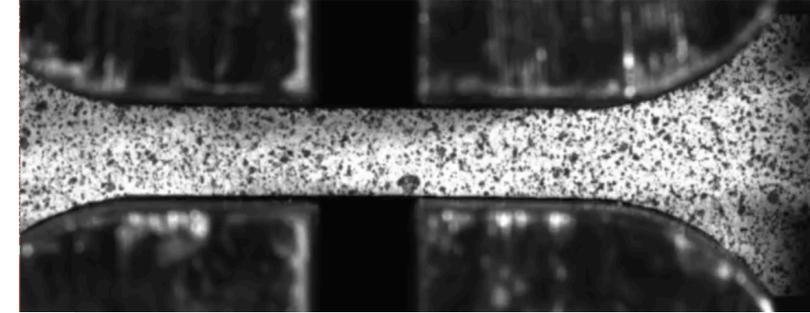


# Dual deformation bands originated and terminated in center of gauge section of D7-1 from Target 9

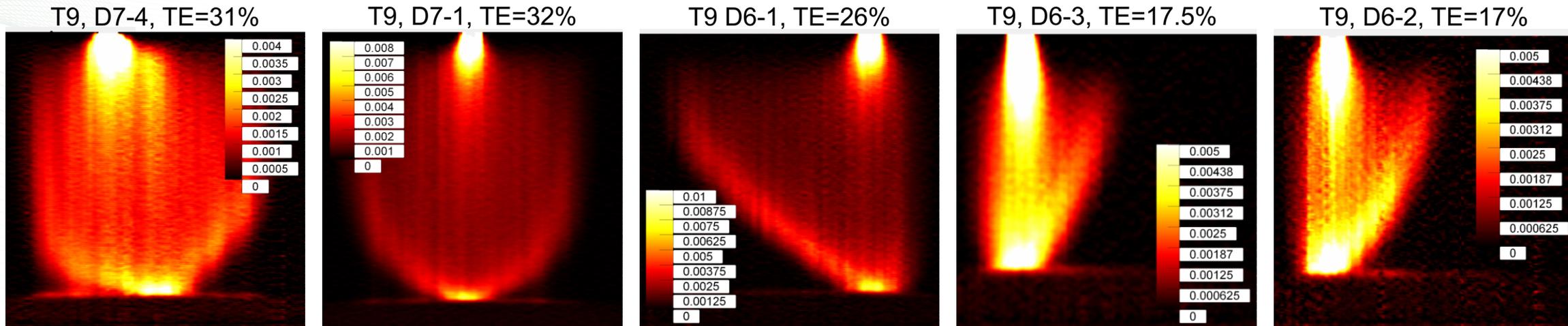
- Two deformation bands originated at the center of the gauge section
- Bands propagated outward and consumed gauge section



Target 9, D7-1 (5.8 dpa), TE = ~33%



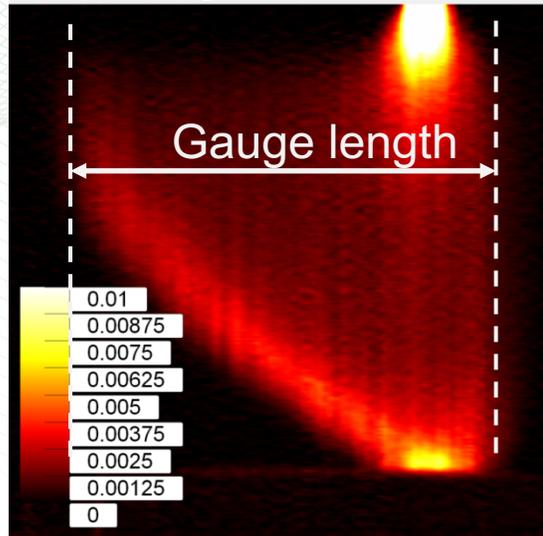
# Examples of deformation behavior of Target 9 specimens



- Several different behaviors lead to significantly different measured properties
- But “traditional” metrics don’t seem to capture the true (ductile) behavior of the material

# What is the best way to interpret abnormal deformation?

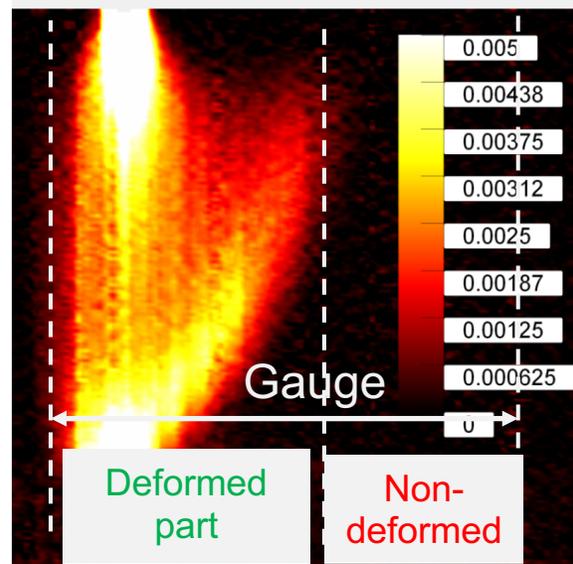
Target 9, D6-1



Band passed along all the gauge (TE~26%).

Therefore, total elongation seems to be correct.

Target 9, D6-2



Band passed only ~60% of the gauge (TE~17%).

Therefore, total elongation value appears to be underestimated.

- DIC revealed specific plastic strain distribution patterns
- For some specimens part of the gauge was not involved in deformation
- **Are we getting the an accurate measure of the actual ductility for the irradiated specimens?**
- **Is there a way to better estimate the actual ductility value?**

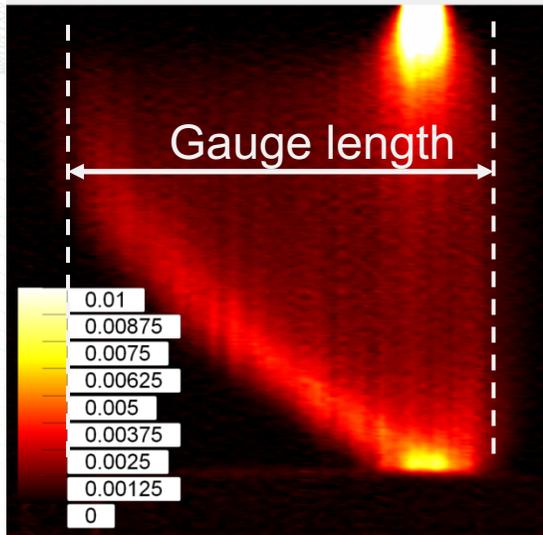
Simplified approach might be just to correct the ductility values, taking into account the non-deformed gauge length:

$$TE[\text{corrected}] \approx TE [\text{observed, 17\%}] * (1/0.6) \approx 28\%.$$

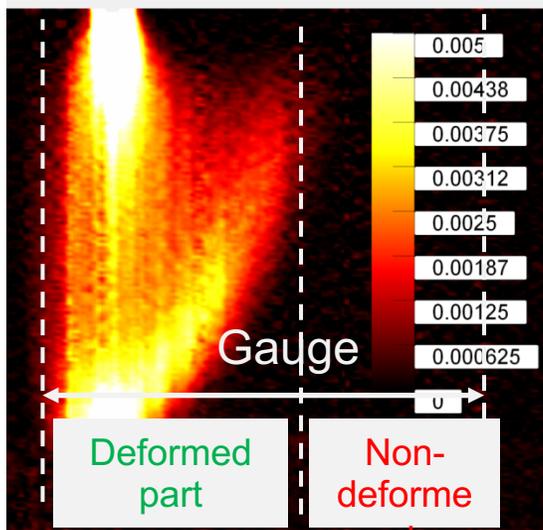
The estimated value is in good agreement with the result for fully-deformed specimens.

# Estimating real ductility using expected uniform elongation

T9D6-1



T9D6-2



- The concept discussed above may be further improved by employing simple constitutive equation analysis.
- During tensile testing, necking occurs when hardening rate becomes less than acting stress:

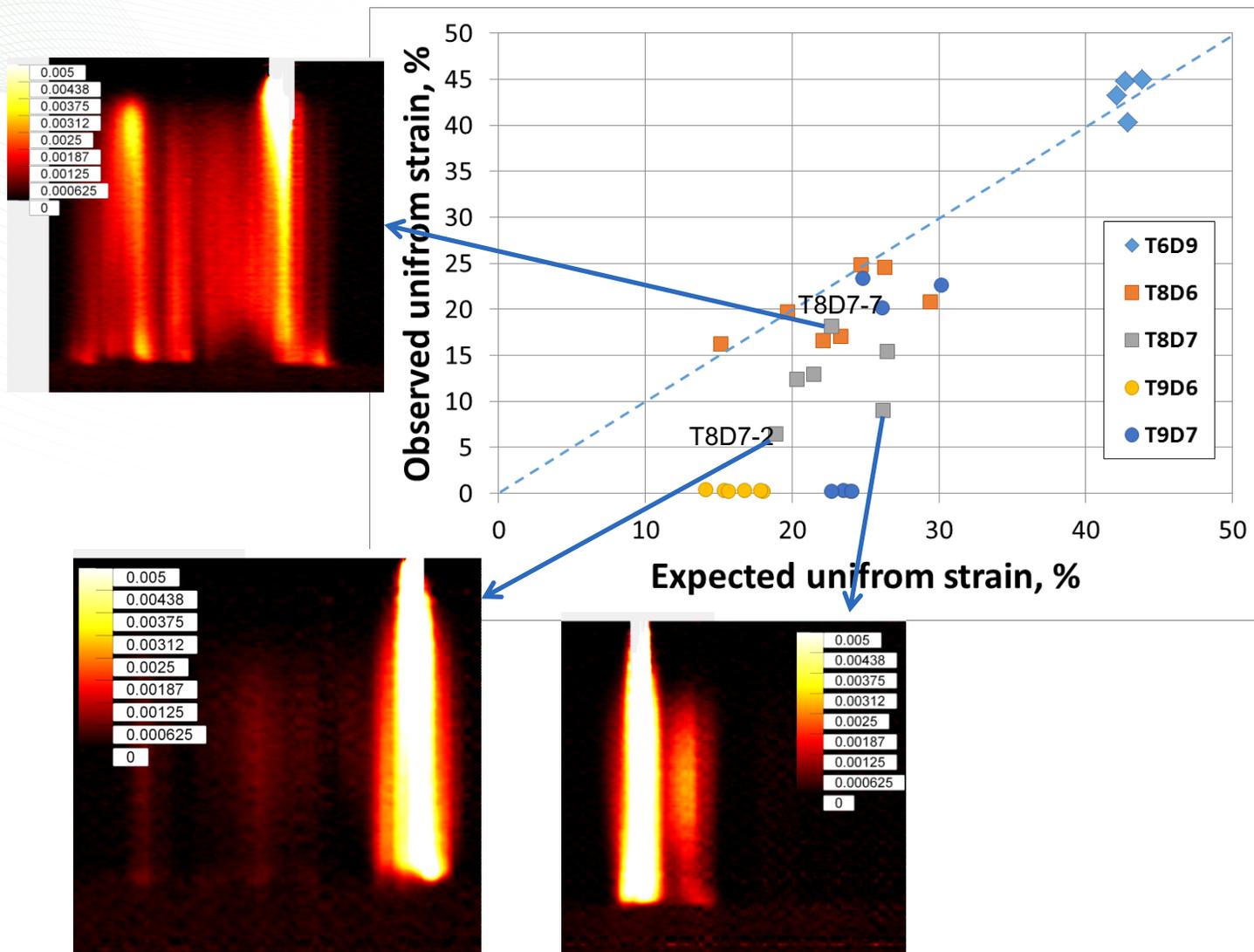
$$d\sigma/d\varepsilon \leq \sigma \quad [1]$$

- Swift equation,  $\sigma = k \times (\varepsilon - \varepsilon_0)^{0.5}$  [2] provide good true stress approximation. Differentiating this equation and using [1] it is easy to show that:

$$\varepsilon_{EUD} = 0.5 (1 - 2\varepsilon_0).$$

- $\varepsilon_{EUD}$  (expected uniform deformation) may be easily estimated using DIC-obtained true stress – true strain curves.

# Expected vs. observed uniform ductility



Experimental and calculated uniform elongation (UE) values were compared for the tested specimens.

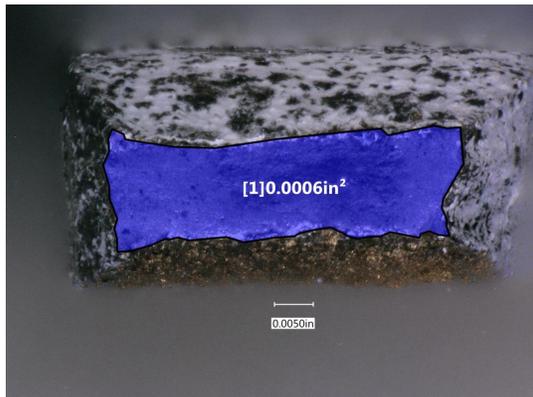
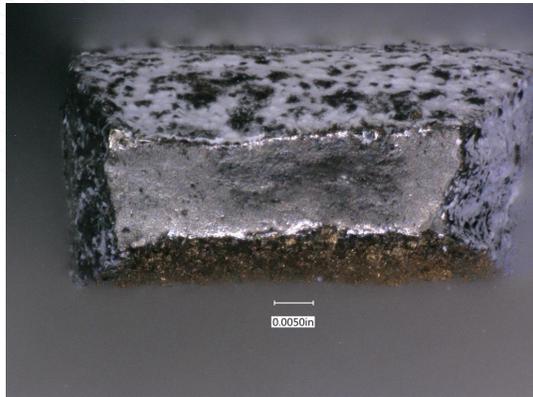
One may see the low-dose (T6D9) specimens demonstrated good agreement between the expected and experimental UE values.

At the same time many specimens showed much smaller UE compared to the expected ones.

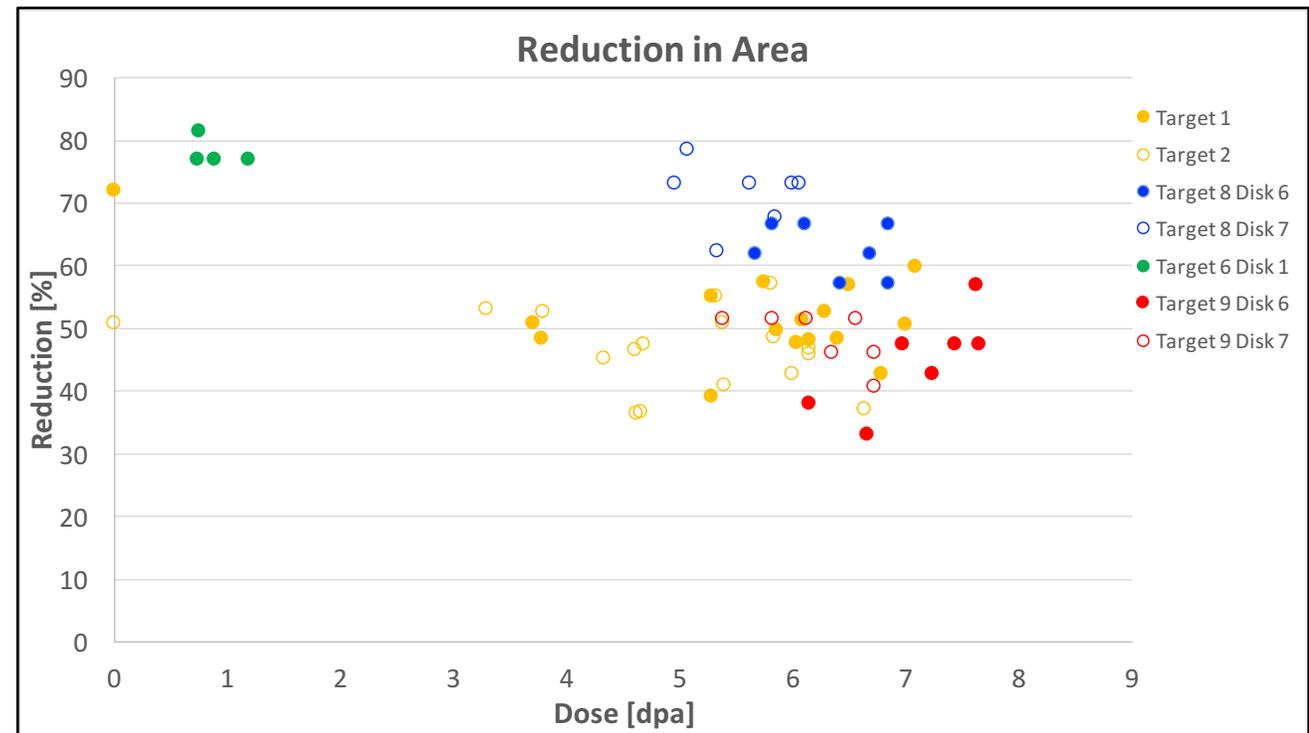
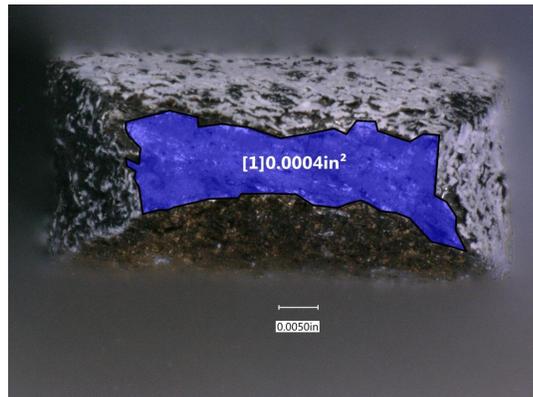
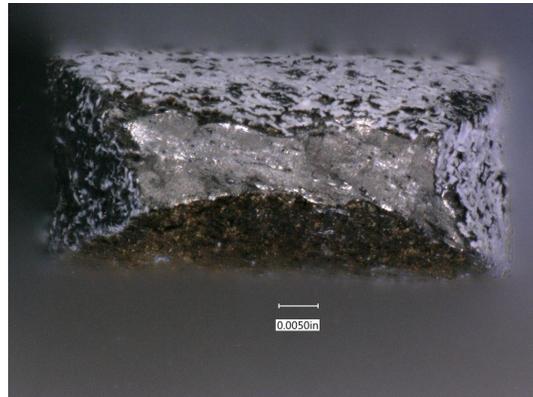
# Reduction in area might be a better indicator of ductility instead of elongation

- Final area was measured for all specimens after testing using digital image analysis

T8 D7-3



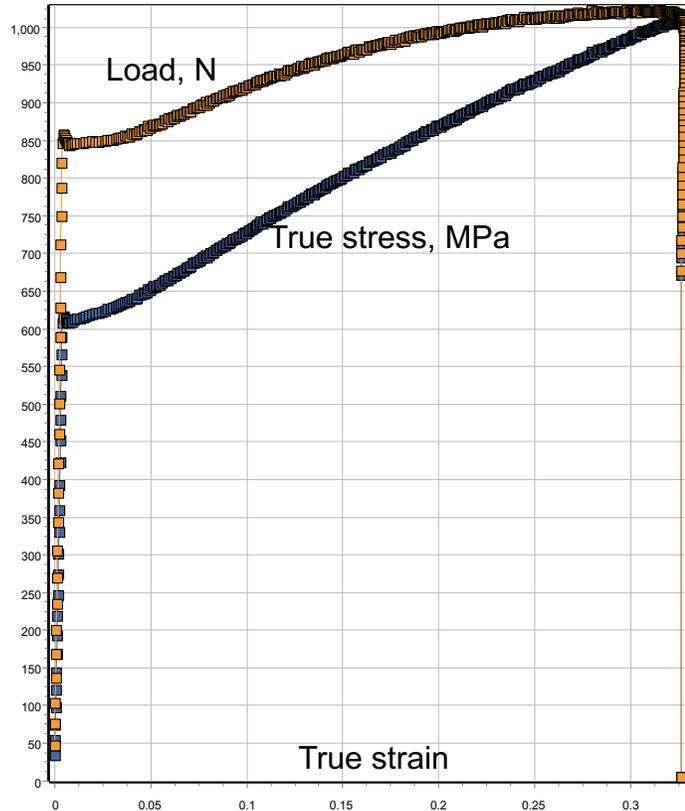
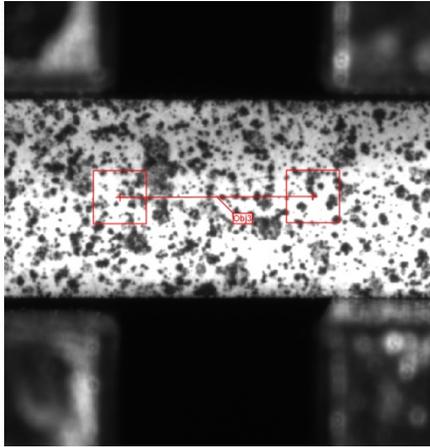
T8 D7-7



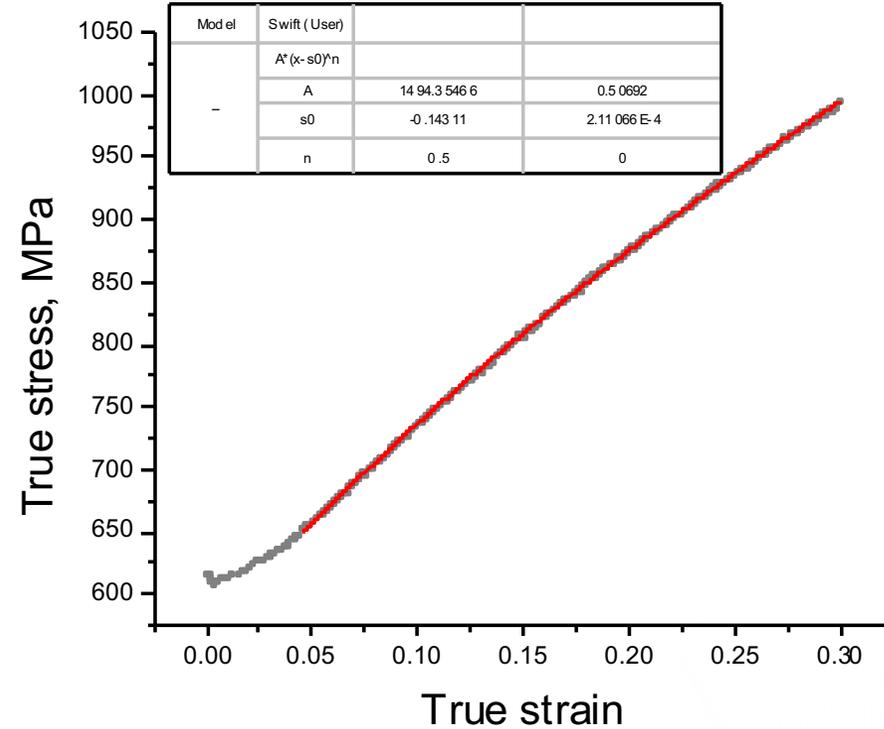
# Summary

- 32 tensile specimens were fabricated and tested from SNS Targets 6, 8, and 9
- Results show 316L retained appreciable ductility (18-24% TE) for doses up to ~7.6 dpa
- Digital Image Correlation analysis showed non-uniform deformation occurred in several specimens
  - For some specimens, only a fraction of the gauge section participated in plastic deformation
- DIC results suggest typical elongation values (UE, TE, and RA) do not appear to accurately reflect the true ductility
  - A more accurate metric is needed

# Strain hardening behavior (T6D1\_1)



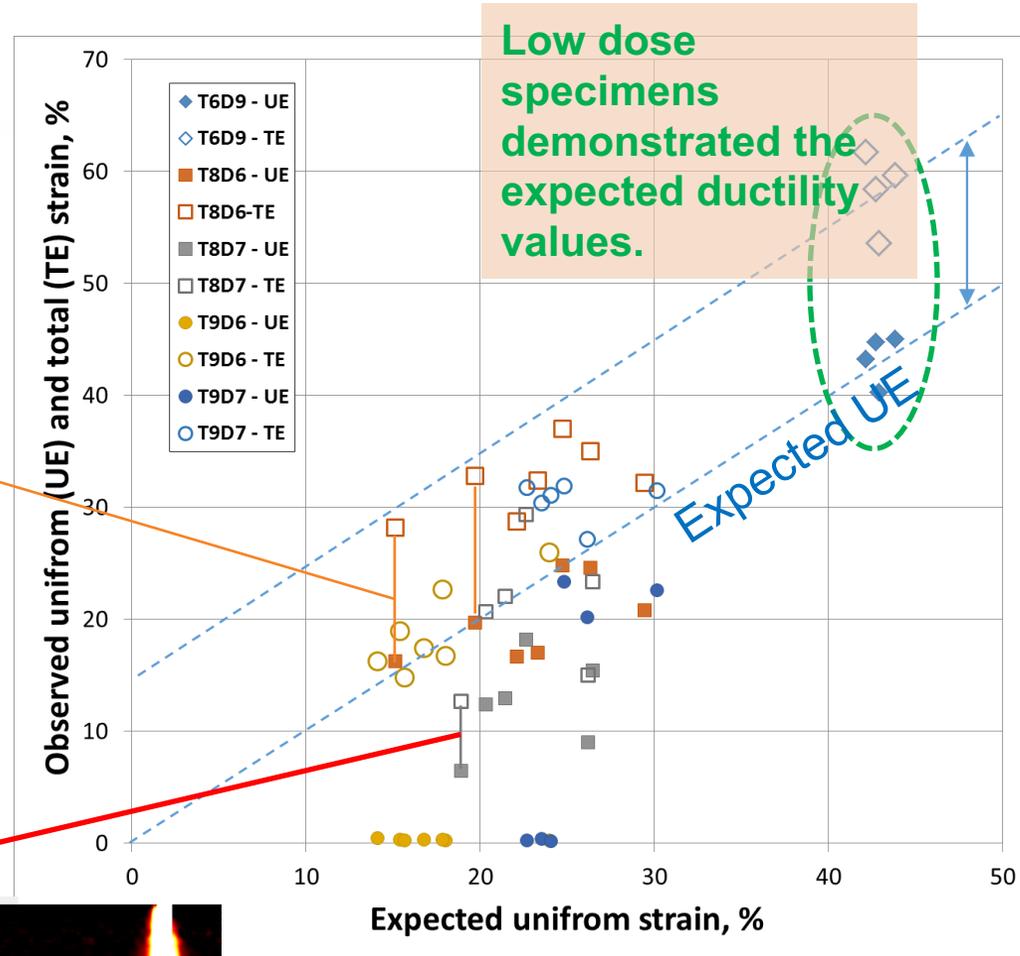
“Usual” Swift equation.



True stress–true strain curves may be calculated for all specimens.

The curves reflect real (acting) plastic deformation behavior and strain hardening.

# Expected vs. observed ductility



Some irradiated specimens fully realized the ductility resource

However, many objects showed very small ductility due to the early strain localization.

